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Randomly branching θ -polymers in two and three dimensions: Average properties and distribution functions

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Motivated by renewed interest in the physics of branched polymers, we present here a detailed characterization of the connectivity and spatial properties of 2 and 3-dimensional single-chain conformations of randomly branching polymers in θ -solvent conditions obtained by Monte Carlo computer simulations. The first part of the work focuses on polymer average properties, like the average polymer spatial size as a function of the total tree mass and the typical length of the average path length on the polymer backbone. In the second part, we move beyond average chain behavior and we discuss the complete distribution functions for tree paths and tree spatial distances, which are shown to obey the classical Redner-des Cloizeaux functional form. Our results were rationalized first by the systematic comparison to a Flory theory for branching polymers and, next, by generalized Fisher-Pincus relationships between scaling exponents of distribution functions. For completeness, the properties of θ -polymers were compared to their ideal (*i.e.*, no volume interactions) as well as good-solvent (*i.e.*, above the θ -point) counterparts. The results presented here complement the recent work performed in our group [A. Rosa and R. Everaers, J. Phys. A: Math. Theor. 49, 345001 (2016), J. Chem. Phys. 145, 164906 (2016), Phys. Rev. E 95, 012117 (2017)] in the context of the scaling properties of branching polymers.

I. INTRODUCTION

Branched polymers or trees represent a fundamental class of polymers whose physics is far more intricate and range of applications far more wider than the familiar example of *linear* polymers [1].

From the theoretical point of view, physical models for randomly branched polymers were introduced to account for the behavior of synthetic as well as biological macromolecules, ranging from star molecules as in the classical work by Zimm and Stockmayer [2] to more recent applications which include: the synthesis of hyperbranched polymers [3, 4], the folding of single-stranded RNA molecules in viral capsids [5–7] and the design of novel soft materials [8] for specific practical scopes as efficient drug delivery [9].

In the more or less close past, there have been considerable efforts [10–14] in trying to predict how the effects of branching on polymer structure are expected to impact on polymer relaxation and dynamics. In particular, our renewed interest in the field of branched polymers has been motivated because of the connection between those and the large scale behavior of unlinked and unknotted ring polymers in concentrated solutions and melt [15–20] and in more generic topologically-constraining environments [21].

From the theoretical point of view, *linear* polymers are described [22, 23] in terms of the expectation value of the square gyration radius $\langle R_g^2(N) \rangle$ of the chain which grows as a power law of the total weight of the polymer, N :

$$\langle R_g^2(N) \rangle \sim N^{2\nu}. \quad (1)$$

ν is the metric scaling exponent of the chain, and it depends on the spatial dimension d and the nature of the solvent surrounding the polymer [23]: in good solvents monomers effectively repel each other and $\nu \approx 3/(d+2)$ for $d \leq 4$ which corresponds to the scaling exponent of the so-called self-avoiding random-walk [24–26], while in bad solvents monomer-monomer attraction is strong enough to fold the polymer into a compact globular state with $\nu = 1/d$. For many solvents, the quality depends on temperature T : at the so-called θ -temperature $T = T_\theta$ repulsion and attraction balance each other almost exactly, thus the chain behaves like under quasi-ideal conditions [27] with $\nu_\theta = \nu_{\text{ideal}} = 1/2$ in $d = 3$.

Because of their more complicate nature, a quantitative description of the structure of branched polymers needs the introduction of additional observables, namely [20, 28]:

1. The average path length, $\langle L(N) \rangle$, between pairs of monomers on the tree as a function of N :

$$\langle L(N) \rangle \sim N^\rho. \quad (2)$$

2. The average branch weight, $\langle N_{\text{br}}(N) \rangle$, as a function of N :

$$\langle N_{\text{br}}(N) \rangle \sim N^\epsilon. \quad (3)$$

3. The mean square spatial distance, $\langle R^2(\ell) \rangle$, between pairs of tree nodes as a function of their mutual path distance, ℓ :

$$\langle R^2(\ell) \rangle \sim \ell^{2\nu_{\text{path}}}. \quad (4)$$

4. The mean contact probability, $\langle p_c(\ell) \rangle$, between pairs of tree nodes as a function of ℓ :

$$\langle p_c(\ell) \rangle \sim \ell^{-\nu_{\text{path}}(d+\theta_{\text{path}})}. \quad (5)$$

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Thus, the scaling exponent ν is complemented by other exponents ρ , ϵ , ν_{path} and θ_{path} which, once again, turn out to depend on spatial dimension d and on the universality class of the system: now, the latter depends not only on the quality of the surrounding solvent [23] as for the linear case, but it also depends on the chain connectivity being quenched or annealed [28–30] and, in solutions of many chains, on the inter-chain polymer-polymer interactions [20]. Two additional relationships between the exponents complete the picture: the obvious $\nu = \rho \nu_{\text{path}}$ and the less trivial $\rho = \epsilon$ [31].

To our knowledge, only very few exact values of these exponents have been reported in the literature. For ideal (*i.e.* without excluded volume effects) tree polymers $\nu_{\text{path}} = \rho = \epsilon = 1/2 = 2\nu$ and $\theta_{\text{path}} = 0$ [2, 28, 32], while for $3d$ single trees in good solvent $\nu = 1/2$ [33]. Otherwise, in the rest of the cases, approximate values for scaling exponents have been worked out by resorting to rather sophisticated numerical or theoretical tools, which the interested reader can be found reported and discussed in the recent review work [34].

By using a suitable combination of Flory theory, scaling arguments and numerical simulations, Rosa and Everaers contributed to characterize the physics of randomly branching polymers by providing predictions for the scaling exponents of Eqs. (1) to (5) for $3d$ single self-avoiding trees in good solvent with annealed and quenched branching statistics [28] and for melts of trees in $2d$ and $3d$ [20]. Later on, they extended [35] the analysis of these simulations by studying the distribution functions for the different observables which contributed to highlight the limits of mean-field-like Flory theory in describing the structural properties of lattice trees [34].

In this article, we add the missing piece to the picture described in these previous works [20, 28, 35] by considering the case of randomly branching polymers with annealed connectivity [28–30] in θ -solvent in $2d$ and $3d$. In particular, we provide here a complete characterization of these specific polymer ensembles by studying their average properties in terms of the observables (1) to (5) and the associated distribution functions, and we compare those to corresponding ensembles of ideal trees and polymers in good solvent conditions. As in works [20, 28], the discussion is guided through the systematic comparison to the predictions (and limitations) of the mean-field-like Flory theory [34].

The paper is organized as follows. In Section II we briefly review the Flory theory for lattice trees, as well as the scaling properties of distribution functions that provide information beyond the mean-field approximation. In Sec. III we describe the numerical model for the branching polymers, in particular the procedure to derive an adequate force field for the θ -polymers, then the algorithm used to perform the Monte Carlo simulations and the methods employed to analyze the output data. In Sec. IV we discuss the main results obtained in this work, while additional plots and Tables are placed in the Appendices at the end of the work. Finally, in Sec. V we

outline the conclusions.

II. THEORETICAL BACKGROUND

A. Flory theory for randomly branching polymers

In spite of being based on rather crude assumptions, Flory theories [34, 36–38] provide remarkably accurate results for the scaling exponents of observables Eqs. (1)–(4) (however (!), as it will be addressed more explicitly below, nothing can be really said about θ_{path} aside from the trivial expectation $\theta_{\text{path}} = 0$). In this Section, we derive these theoretical exponents for interacting polymers in good and θ -solvent conditions. In this way, we are able to explore the differences between the scaling exponents obtained from the simulations and those expected from the mean-field Flory theory.

For randomly branching polymers in a generic solvent in d -dimensions and neglecting numerical prefactors, the Flory free energy (\mathcal{F}) is formulated as a balance of three terms:

$$\frac{\mathcal{F}}{k_B T} \sim \frac{R^2}{\ell_K L} + \frac{L^2}{\ell_K^2 N} + v_p \frac{N^p}{R^{(p-1)d}}. \quad (6)$$

The first and second terms account for the entropic contributions coming from stretching the polymer [39] between any two ends and from branching statistics [40, 41], respectively. The third term is the general expression for the (solvent-mediated) p -body interaction between monomers derived from the classical virial expansion [34, 38]. Thus, under good solvent conditions $p = 2$ describes the volume exclusion resulting from 2-body interactions. Instead, at the θ -point, the latter term vanishes and the effective interactions between monomers are mediated by the 3-body term with $p = 3$.

Minimization of Eq. (6) with respect to L and R leads to:

$$L \sim (\ell_K R^2 N)^{1/3}, \quad (7)$$

$$R \sim \ell_K N^{(3p+1)/(4+3(p-1)d)}, \quad (8)$$

which imply the following results for the scaling exponents:

$$\begin{aligned} \nu &= \frac{3p+1}{4+3(p-1)d}, \\ \rho = \epsilon &= \frac{2(p+1) + (p-1)d}{4+3(p-1)d}, \\ \nu_{\text{path}} = \frac{\nu}{\rho} &= \frac{3p+1}{2(p+1) + (p-1)d}. \end{aligned} \quad (9)$$

For single randomly branching polymers in good solvent

$p = 2$, so Eqs. (9) give:

$$\begin{aligned}\nu &= \frac{7}{4 + 3d}, \\ \rho = \epsilon &= \frac{6 + d}{4 + 3d}, \\ \nu_{\text{path}} &= \frac{\nu}{\rho} = \frac{7}{6 + d},\end{aligned}\quad (10)$$

while in θ -solvent $p = 3$ and

$$\begin{aligned}\nu &= \frac{5}{2 + 3d}, \\ \rho = \epsilon &= \frac{4 + d}{2 + 3d}, \\ \nu_{\text{path}} &= \frac{\nu}{\rho} = \frac{5}{4 + d}.\end{aligned}\quad (11)$$

Finally, because Flory theories neglect chain correlations *by construction*, $\theta_{\text{path}} = 0$. It is particularly worth noticing that, contrarily to linear chains, for branched polymers $\nu_{\theta}(d = 3) > \nu_{\text{ideal}} = 1/4$, which can be intuitively understood in terms of the impossibility to design an object with fractal dimension higher than the embedding space. Nonetheless, the present work will show that some observables, like the asymptotic branching probability, remain fairly the same in ideal and θ -conditions, while many others describing branch weight and conformational statistics will differ considerably as the result of volume interactions.

In this work we will mainly focus on $2d$ and $3d$ θ -polymers and we will compare the predictions of Eqs. (11) with the results of Monte Carlo computer simulations. For completeness, we discuss those together with the properties of ideal as well as randomly branching polymers in good solvent (Eqs. (10)).

B. Beyond Flory theory: Distribution functions

Even though Flory theory provides a useful framework to study the scaling properties of branching polymers, its mean-field-like nature is based on rather crude assumptions [42, 43], *in primis* the Gaussian functional form for quantifying the entropy in Eq. (6).

An example of the limitations of the Flory theory is given by the well known example of the end-to-end distribution function, $p(\vec{r}_{ee})$, for d -dimensional self-avoiding linear (*i.e.*, unbranched) polymers in good solvent conditions. It turns out in fact that, in the large ($N \rightarrow \infty$) polymerization limit $p(\vec{r}_{ee})$ obeys the following scaling ansatz:

$$p(\vec{r}_{ee}) = \frac{1}{\langle R^2(N) \rangle^{d/2}} q\left(\frac{\vec{r}_{ee}}{\langle R^2(N) \rangle^{1/2}}\right), \quad (12)$$

where $\langle R^2(N) \rangle \sim N^{2\nu}$ and

$$q(x) = C x^{\theta} \exp\left[-(Kx)^t\right] \quad (13)$$

satisfies the so-called Redner-des Cloizeaux (RdC) functional form [44–46]. The two constants C and K can be computed by imposing that: (1) $q(x)$ normalizes to 1 and (2) the second moment constitutes the only scaling length [35, 47]. With these constraints we get easily the following analytical expressions:

$$C = t \frac{\Gamma(1 + \frac{d}{2}) \Gamma(\frac{d+\theta}{2} (\frac{2+d+\theta}{t}))}{d \pi^{d/2} \Gamma(\frac{2+d+\theta}{2} (\frac{d+\theta}{t}))}, \quad (14)$$

$$K^2 = \frac{\Gamma(\frac{2+d+\theta}{t})}{\Gamma(\frac{d+\theta}{t})}. \quad (15)$$

Here, $\Gamma = \Gamma(x)$ is the standard Euler's Γ -function. Thus, the knowledge of the pair of exponents (θ, t) is enough to reconstruct the full distribution function in the asymptotic limit of large N . Interestingly, the two exponents θ and t are not completely independent and are related to the other scaling exponents. Thus, (a) the “mechanical” Fisher-Pincus [48, 49] relationship imposes that:

$$t = \frac{1}{1 - \nu}, \quad (16)$$

while (b):

$$\theta = \frac{\gamma - 1}{\nu} \quad (17)$$

constitutes a sort of measure of the “entropy” of the walks since the exponent γ appears [43, 50] in the partition function $\mathcal{Z}_N \sim N^{\gamma-1} \mu^N$ of the walks. Notice that, with $\nu = 1/2$ and $\gamma = 1$ (or, $\theta = 0$) the RdC function reduces to the classical Gaussian function describing ideal polymers [22, 23].

As shown in the recent work [35] by our group, the RdC formalism can be easily generalized to describe the scaling behavior of the distribution functions of the observables considered in Eqs. (1)–(5).

To fix the ideas, we focus on the following functions: (1) the distribution function, $p_N(\ell)$, of linear paths of contour length ℓ on polymers of weight N ; (2) the distribution function, $p_N(\vec{r}|\ell)$, of the end-to-end spatial distances for linear paths of contour length ℓ , and (3) the distribution function of spatial distances between pairs of nodes, $p_N(\vec{r})$. The three functions are not independent, as they satisfy the obvious convolution-like identity:

$$p_N(\vec{r}) = \int_0^\infty p_N(\vec{r}|\ell) p_N(\ell) d\vec{\ell}. \quad (18)$$

Asymptotically, these functions display universal behaviors and, respectively, can be expressed by the following scaling forms:

$$p_N(\ell) = \frac{1}{\langle L(N) \rangle} q\left(\frac{\ell}{\langle L(N) \rangle}\right), \quad (19)$$

$$p_N(\vec{r}|\ell) = \frac{1}{\langle R^2(\ell, N) \rangle^{d/2}} q\left(\frac{|\vec{r}|}{\langle R^2(\ell, N) \rangle^{1/2}}\right), \quad (20)$$

$$p_N(\vec{r}) = \frac{1}{(2 \langle R_g^2(N) \rangle)^{d/2}} q\left(\frac{|\vec{r}|}{(2 \langle R_g^2(N) \rangle)^{1/2}}\right). \quad (21)$$

In all cases described by Eqs. (19)-(21), the function $q(x)$ obeys the RdC functional form Eq. (13) with *novel* pairs of exponents called, respectively, (θ_ℓ, t_ℓ) , $(\theta_{\text{path}}, t_{\text{path}})$ and $(\theta_{\text{tree}}, t_{\text{tree}})$ [35], whose knowledge appears of fundamental importance for understanding the physics of branching polymers in different solvent conditions.

As for the functions in Eqs. (20) and (21), corresponding constants C and K are given by expressions analogous to Eqs. (14) and (15). As for Eq. (19) instead, the obvious normalization to 1 of $q(x)$ is accompanied by imposing that the *first* (rather than the second [35]) moment constitutes the only scaling length, thus implying the following expressions:

$$C = t_\ell \frac{\Gamma^{\theta_\ell+1}((\theta_\ell+2)/t_\ell)}{\Gamma^{\theta_\ell+2}((\theta_\ell+1)/t_\ell)}, \quad (22)$$

$$K = \frac{\Gamma((\theta_\ell+2)/t_\ell)}{\Gamma((\theta_\ell+1)/t_\ell)}. \quad (23)$$

In analogy to the example of self-avoiding linear polymers, the different pairs of exponents (θ, t) are quantitatively related to the exponents $(\nu, \rho, \nu_{\text{path}})$ characterizing the scaling behavior of the trees average properties through generalized Fisher-Pincus [48, 49] relations, as documented first in Ref. [35]. Thus the exponent ρ , which describes the scaling of the average path length, can be used to compute the exponents (θ_ℓ, t_ℓ) describing the distribution of linear paths of contour length, $p_N(\ell)$,

$$\theta_\ell = \frac{1}{\rho} - 1, \quad (24)$$

$$t_\ell = \frac{1}{1 - \rho}. \quad (25)$$

Similarly, the exponent t_{path} is related to ν_{path} :

$$t_{\text{path}} = \frac{1}{1 - \nu_{\text{path}}}, \quad (26)$$

while, interestingly, θ_{path} (which describes the decay of the mean contact probability, Eq. (5)) is independent by the others and, then, it constitutes a genuinely *novel* exponent. Finally, the exponents for the distribution of spatial distances between pairs of nodes, $p_N(\vec{r})$, are related to the metric scaling exponent of the polymer chain ν and θ_{path} by:

$$\theta_{\text{tree}} = \min \left(\theta_{\text{path}}, \frac{1}{\nu} - d \right), \quad (27)$$

$$t_{\text{tree}} = \frac{1}{1 - \nu}. \quad (28)$$

The interested reader can find a complete account on the mathematical derivation and physical meaning of these relations in Ref. [35].

III. MODEL AND METHODS

In this article we generalize the polymer model and Monte Carlo algorithm described in the former works [20,

28], the details relevant here being summarized respectively in Sec. III A and Sec. III B. The main novelty of this work, namely how to model θ -polymers, is explained in detail in Sec. III C. Finally, notation, algorithms and methods (also inspired by works [20, 28]) employed in the characterization of polymer conformations and estimation of scaling exponents are described in Sec. III D.

A. Branching polymers on the square and cubic lattices

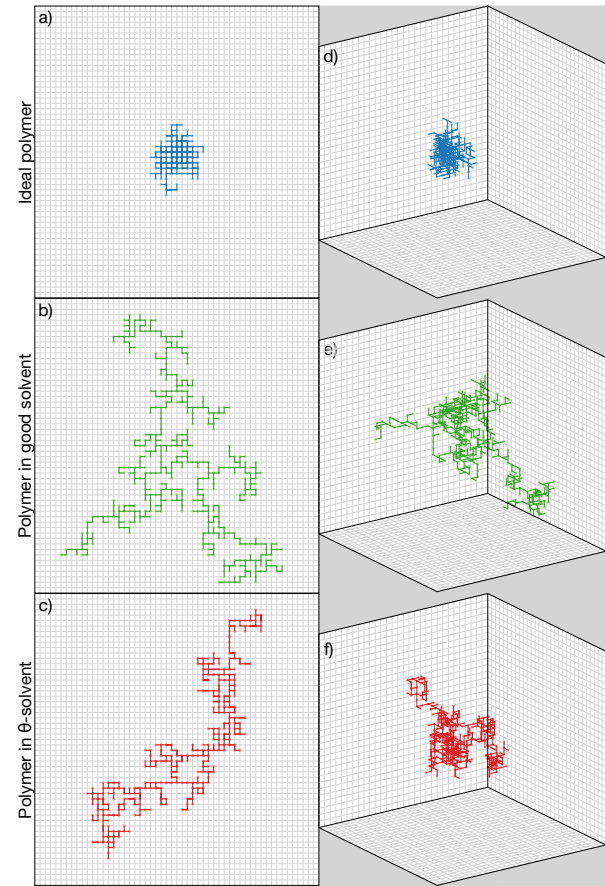


FIG. 1. Representative conformations for $2d$ (l.h.s. panels) and $3d$ (r.h.s. panels) randomly branching polymers from different ensembles: (a, d) ideal polymers, (b, e) single randomly branching polymers in good solvent, (c, f) single randomly branching polymers in θ -solvent. For interacting polymers, the size of the simulation box is typically much larger than the lattice portion shown here so to guarantee the dilute regime.

We model randomly branching polymers as *loop-free* trees on the square ($d = 2$) and cubic ($d = 3$) lattices with periodic boundary conditions. For self-interacting polymers (see Sec. III B) the simulation box was chosen large enough to ensure that we have been working in the dilute regime.

To fix the notation, we consider polymers or trees consisting of a branched structure in which $N + 1$ nodes are

connected by N Kuhn [23] segments of unit length ℓ_K and unit mass m . For simplicity and without any loss of generality [20, 28] we limit the maximal functionality of each node (corresponding to the number of bonds protruding from the node) to $f \leq 3$.

We stress that we consider trees with *annealed* connectivity [28], which means that the location of the branching points undergoes thermal fluctuations as the result of the coupling to an external control parameter. It turns out that this is very different from the ensemble where connectivity is kept *quenched* as in the case of chemically-synthesized polymers, *branching* polymers with annealed connectivity and *branched* polymers with quenched connectivity belonging to different universality classes [28–30]. Single representative conformations for each of the different polymer ensembles considered in this work are illustrated in Figure 1.

B. Monte Carlo computer simulations

1. The algorithm

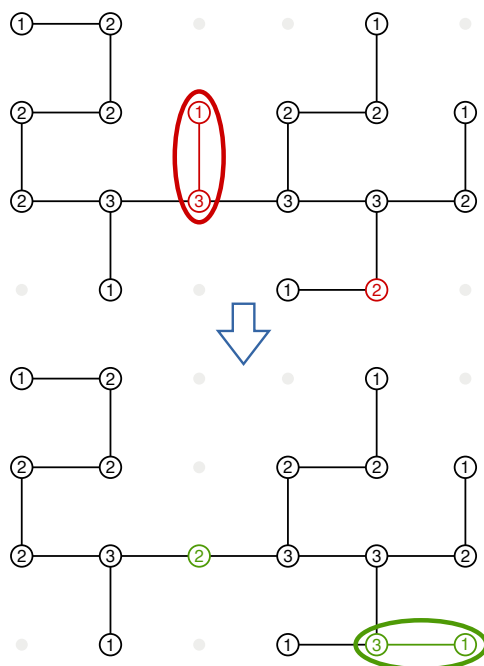


FIG. 2. Schematic drawing describing the version of the amoeba algorithm [51] employed here. Each node of the tree is identified by its corresponding functionality, f . (Top) The red nodes indicate the leaf chosen to be cut and the node where it will be pasted. (Bottom) The new polymer configuration obtained, with corresponding updated links and nodes functionalities.

Monte Carlo simulations of randomly branching polymers are performed according to a slightly modified version of the so-called “amoeba” algorithm by Seitz and

Klein [51]. In this algorithm, each new configuration is generated from the previous one by randomly cutting a leaf from the tree and then reconnecting it randomly to one of the other nodes with functionality $f \leq 2$, thus constraining the single-node functionality to be not larger than $f = 3$ (see Sec. III A). A schematic example of this procedure is shown in Fig. 2.

Each polymer configuration is characterized by the position of all its nodes, $\Gamma = \{\vec{r}_1, \dots, \vec{r}_{N+1}\}$, and their connectivity, \mathcal{G} . Both Γ and \mathcal{G} are modified in a trial move of the amoeba algorithm. This move will be accepted with probability given by the standard Metropolis [52] algorithm accounting for detailed balance:

$$\text{acc}_{i \rightarrow f} = \min \left\{ 1, \frac{n_1(i)}{n_1(f)} e^{-\frac{1}{k_B T} [\mathcal{H}(\Gamma_f, \mathcal{G}_f) - \mathcal{H}(\Gamma_i, \mathcal{G}_i)]} \right\}, \quad (29)$$

where $1/k_B T$ is the Boltzmann factor, n_1 is the (initial and final) total number of nodes with functionality $f = 1$ and $\mathcal{H}(\Gamma, \mathcal{G})$ is the total interaction Hamiltonian (described in Sec. III C) between nodes.

2. Polymer equilibration and statistics

Single tree conformations are initially prepared as ideal random walks on the lattice. The equilibration of our systems was checked by monitoring (see Fig. 15 in the Appendix) that (1) the ensemble-average square gyration radius of the polymer, $\langle R_g^2(t_{MC}) \rangle$, and (2) the ensemble-average number of branching nodes, $\langle n_3(t_{MC}) \rangle$, as functions of the Monte Carlo “time” steps t_{MC} both reach equilibrium values. The total number, M , of statistically independent tree conformations used for the averages is: $M = 100$ for $3 \leq N \leq 10$; $M = 1000$ for $20 \leq N \leq 900$; $M = 2000$ for $N = 1800$.

C. Interaction Hamiltonian for branching polymers

The total energy of the system is given by the sum of two contributions, one ideal and one due to the interactions between nodes [20, 28]:

$$\mathcal{H}(\Gamma, \mathcal{G}) = \mathcal{H}_{\text{id}}(\mathcal{G}) + \mathcal{H}_{\text{int}}(\Gamma). \quad (30)$$

The ideal contribution, $\mathcal{H}_{\text{id}}(\mathcal{G})$, controls the connectivity of the chain and it is expressed in terms of the coupling between the chemical potential of branching points, μ_{br} , and the total number of 3-functional (branching) nodes in the polymer, $n_3(\mathcal{G})$:

$$\frac{\mathcal{H}_{\text{id}}(\mathcal{G})}{k_B T} = \mu_{\text{br}} n_3(\mathcal{G}). \quad (31)$$

For consistency with previous works [20, 28] based on the connection [18] between melt of ring polymers and melt of randomly branching polymers and without loss of generality, the value μ_{br} was chosen equal to -2.0 which

thus fixes the average fraction of branching points to ≈ 0.4 , see Fig. 4.

The interaction term, $\mathcal{H}_{\text{int}}(\Gamma)$, between tree nodes is described as the sum of 2- and 3-body interactions,

$$\frac{\mathcal{H}_{\text{int}}(\Gamma)}{k_{\text{B}}T} = \alpha_2 \sum_{j \in \text{lattice sites}} \kappa_j^2 + \alpha_3 \sum_{j \in \text{lattice sites}} \kappa_j^3, \quad (32)$$

where κ_j is the total number of Kuhn segments inside the elementary cell centered at the lattice site j . An expression like Eq. (32) with $\alpha_2 > 0$ and $\alpha_3 = 0$ was already considered in Refs. [20, 28] for modelling single branching polymers in good solvent or branching polymers in melt, as net 2-body repulsive interactions are known to dominate polymer behavior in these regimes [34, 38].

For polymers at the θ -point, repulsive interactions are compensated by a monomer-monomer attraction at short spatial separations. For *linear* polymers this has the important implications that their behavior is quasi-ideal, *i.e.* the scaling exponent $\nu \approx 1/2$ [23]. To properly model branching polymers in θ -conditions we have then balanced the two terms in Eq. (32): an attractive 2-body interaction ($\alpha_2 < 0$) is needed in order to overcome the volume exclusion, which is taken into account through the repulsive 3-body interaction ($\alpha_3 > 0$).

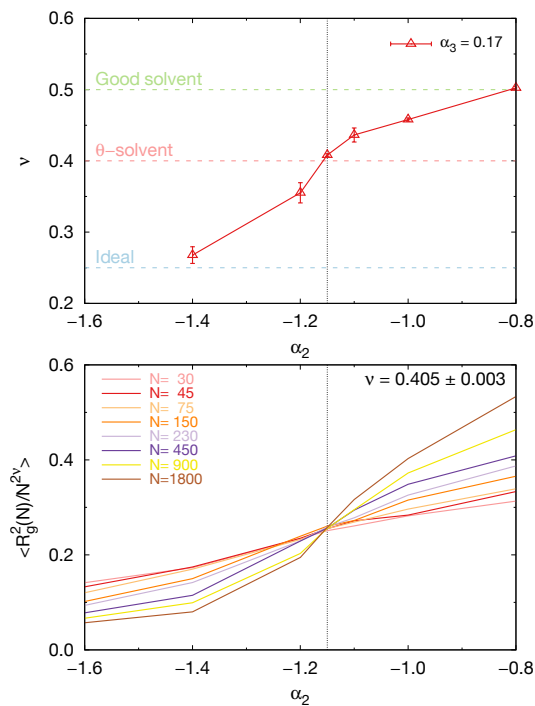


FIG. 3. (Top) Scaling exponent ν for 3d branching polymers calculated for different values of the 2-body interaction parameter, α_2 , with given 3-body interaction parameter $\alpha_3 = 0.17$. θ -solvent conditions correspond to the value $\alpha_2 = -1.15$ where the red curve intercepts the straight line (in red) $\nu = 0.405 \pm 0.003$. (Bottom) Corresponding normalized mean square gyration radii, $\langle R_g^2(N) \rangle / N^{2\nu}$, with $\nu = 0.405 \pm 0.003$.

Finding the adequate values for α_2 and α_3 was done in

the following way, as schematically illustrated in Fig. 3 (top). We considered 3d branching polymers and we found that for the pair $(\alpha_2, \alpha_3) = (0.0, 0.17)$ we observe the expected scaling behavior of branching polymers in good solvent conditions, $\langle R_g^2(N) \rangle \sim N^{2\nu} \sim N^1$ [33]. Then, at fixed α_3 , we decreased progressively $\alpha_2 < 0$ and measured the corresponding scaling exponent ν , which diminishes accordingly. At the value $\alpha_2 = -1.15$ we found that the exponent is equal to $\nu = 0.405 \pm 0.003$, which is in very good agreement with the most accurate reference estimate $\nu_{\text{ref}} = 0.400 \pm 0.005$ obtained by Madras and Janse van Rensburg from Monte Carlo computer simulations of 3-dimensional θ -trees [34, 53]. For even lower values of α_2 , the polymers are found to collapse into quasi-ideal conformations with $\nu \approx 1/4$. Fig. 3 (bottom) illustrates nicely the effectiveness of this procedure by plotting the quantity $\langle R_g^2(N) \rangle / N^{2\nu}$ with $\nu = 0.405 \pm 0.003$ as a function of α_2 and for different N . For $\alpha_2 = -1.15$ all curves intersect each other at the same point where finite- N effects appear negligible. Conversely, for smaller (respectively, larger) values of α_3 the different curves show a decreasing (resp., increasing) trend at increasing values of N .

Importantly, while the value of ν for our three-dimensional θ -trees was obtained by fitting the interaction parameters of Eq. (32) so to reproduce the numerical result by Madras and Janse van Rensburg, we anticipate (see Sec. IV A) that the same parameters can be employed to model θ -trees in two dimensions without additional fine tuning. This finding demonstrates that our lattice model describes correctly the physics of randomly branching θ -polymers.

D. Analysis of polymer conformations

Equilibrated polymer conformations obtained by Monte Carlo computer simulations were analyzed by following closely the definitions, algorithms and tools described in Refs. [20, 28, 35]. In particular, for characterizing the scaling behaviors of trees connectivity and spatial structure we adopt here the same terminology of these papers so, to avoid unnecessary repetition, we have recapitulated the complete list of observables in Table I. Then, in Sec. III D 1 we have briefly described the so-called “burning” algorithm necessary, in particular, to extract information on the observables quantifying trees connectivity. Finally, scaling exponents and trees asymptotic properties are estimated by the finite-size scaling analysis presented in Sec. III D 2. The reader interested in more details and results concerning other polymer ensembles is invited to look into publications [20, 28, 35].

1. Analysis of nodes connectivity via burning

In order to analyze the node-to-node connectivity we have resorted to a close variant of the “burning” algo-

TABLE I. List of observables and distribution functions considered in this work, with description and reference to the corresponding figures. These quantities have been used to characterize other trees ensembles, see Refs. [20, 28, 35].

Notation	Description	Figure
$\langle n_3(N) \rangle$	Average fraction of 3-functional nodes as a function of the total tree weight N	4
$\langle L(N) \rangle$	Average path length between pairs of nodes as a function of the total tree weight N	5
$\langle \delta \ell_{\text{center}}(N) \rangle$	Average path distance of nodes from the central node as a function of the total tree weight N	5
$\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$	Average <i>longest</i> path distance of nodes from the central node as a function of the total tree weight N	5
$\langle N_{\text{br}}(\delta \ell_{\text{root}}^{\text{max}}) \rangle$	Average branch weight as a function of the longest path to the branch root $\delta \ell_{\text{root}}^{\text{max}}$	6
$\langle N_{\text{center}}(\delta \ell_{\text{center}}) \rangle$	Average branch weight of paths whose distance from the tree center does not exceed $\delta \ell_{\text{center}}$	6
$\langle N_{\text{br}}(N) \rangle$	Average branch weight as a function of the total tree weight N	7
$\langle R^2(\ell = \langle L(N) \rangle) \rangle$	Mean-square end-to-end spatial distance for paths of contour length $\ell = \langle L(N) \rangle$	8
$\langle R^2(L_{\text{max}}(N)) \rangle$	Mean-square end-to-end spatial distance of the longest paths	8
$\langle R^2(\ell, N) \rangle$	Mean-square end-to-end spatial distance for paths of contour length ℓ	9
$\langle p_c(\ell, N) \rangle$	Mean closure probability for paths of contour length ℓ	9
$\langle R_g^2(N) \rangle$	Mean-square tree gyration radius as a function of the total tree weight N	10
$\langle \Lambda_{1,2,3}^2(N) \rangle$	Mean-square eigenvalues of the tree gyration tensor as a function of the total tree weight N	11
$p_N(\ell)$	Probability distribution function of tree paths of total length ℓ	12
$p_N(\vec{r} \ell)$	Probability distribution function of end-to-end vectors \vec{r} between pairs of nodes of total path length ℓ	13
$p_N(\vec{r})$	Probability distribution function of end-to-end vectors \vec{r} between tree nodes	14

rithm originally proposed for percolation clusters [54]. Essentially, this algorithm consists of two steps:

1. An inward step, during which the polymer graph is “burned” from outside to inside by removal of all tips (the nodes with functionality $f = 1$) in order to obtain a smaller tree to which the algorithm is then applied recursively. The procedure stops when only one single node is left (the central node of the original tree).
2. An outward step, consisting in the advancement from the center to the periphery.

In the first step one collects information about the mass and shape of branches, while in the second one reconstructs the distances of nodes from the centre of the tree. The unique minimal path ℓ_{ij} between any pairs of tree nodes i and j is obtained by modifying the burning algorithm ensuring that it does not proceed inward of node i and node j . As our trees contain no loops *by construction*, the procedure ends with a single linear polymer of contour distance = ℓ_{ij} . The interested reader may find a more extensive illustration of the burning procedure in Ref. [28].

2. Estimating scaling exponents: Average properties

Accurate evaluation of scaling exponents for the different observables Eqs. (1)-(5) is non-trivial, as numerical procedures are typically plagued by finite- N effects [31]. In order to overcome this issue, we resort to the numerical strategy employed in the former works [20, 28, 35] dedicated to branching polymers.

Let O be a generic observable which depends on polymer size N so that $\langle O(N) \rangle \sim N^\gamma$ for large N where γ is the corresponding scaling exponent. A first estimate of

the exponent $\gamma = \gamma_1 \pm \delta\gamma_1$ with corresponding statistical error is obtained by best fit of $\log \langle O(N) \rangle$ vs. $\log N$ for $N \geq 450$ to the straight line:

$$\log \langle O(N) \rangle = a + \gamma_1 \log N. \quad (33)$$

Then, in order to account for finite- N effects and thus estimate systematic errors we best fit the data for the full range $N \geq 10$ to the modified function:

$$\log \langle O(N) \rangle = a + bN^{-\Delta} + \gamma_2 \log N, \quad (34)$$

which contains a proper correction-to-scaling [31] term. In practice, instead of solving the non-linear fit with parameters (a , b , Δ and γ_2), we have linearized Eq. (34) around some Δ_0

$$\log \langle O(N) \rangle = a + bN^{-\Delta_0} + \gamma_2 \log N + b(\Delta_0 - \Delta)N^{-\Delta_0} \log N, \quad (35)$$

and linear fit the data accordingly by using different Δ_0 values [28]. Thus, the final estimate for the scaling exponent $\gamma = \gamma_2 \pm \delta\gamma_2$ comes from the fit whose Δ_0 value makes the term $N^{-\Delta_0} \log N$ vanishing.

The quality of both fit procedures Eqs. (33) and (35) is checked by means of standard statistical analysis [55]. The fit is deemed to be reliable when the normalized χ -square, $\tilde{\chi}^2 \equiv \frac{\chi^2}{D-f} \approx 1$. χ^2 is calculated by minimizing the weighted square deviation between the data and the model, and $D - f$ is the number of degrees of freedom, calculated as the difference between the number of data points (D) and the number of fit parameters (f). The corresponding $\mathcal{Q}(D-f, \chi^2)$ -values provide a quantitative indicator for the likelihood that χ^2 should exceed the observed value, if the model were correct [55]. The results of all fits are reported together with the corresponding errors, $\tilde{\chi}^2$ and \mathcal{Q} values in Tables V and VII in the Appendix. The reader will notice that there are a few cases

TABLE II. Final estimated for the scaling exponents of the lattice trees ensembles considered in this work.

	2-dimensions					
	Ideal polymer		Good solvent		θ -solvent	
	Flory	Simulations	Flory	Simulations	Flory	Simulations
ρ	0.5	0.48 ± 0.06	0.8	0.739 ± 0.016	0.75	0.711 ± 0.016
ϵ	0.5	0.49 ± 0.04	0.8	0.743 ± 0.004	0.75	0.720 ± 0.011
ν	0.25	0.24 ± 0.07	0.7	0.624 ± 0.005	0.625	0.61 ± 0.10
ν_{path}	0.5	0.51 ± 0.02	0.875	0.836 ± 0.009	0.833	0.88 ± 0.04
θ_{path}	-	-0.30 ± 0.14	-	1.6 ± 0.2	-	1.6 ± 0.3

	3-dimensions					
	Ideal polymer ^a		Good solvent ^a		θ -solvent	
	Flory	Simulations	Flory	Simulations	Flory	Simulations
ρ	0.50	0.49 ± 0.04	0.692	0.64 ± 0.02	0.636	0.585 ± 0.018
ϵ	0.50	0.536 ± 0.007	0.692	0.655 ± 0.009	0.636	0.591 ± 0.011
ν	0.25	0.25 ± 0.02	0.538	0.48 ± 0.04	0.455	0.405 ± 0.003
ν_{path}	0.50	0.509 ± 0.009	0.778	0.74 ± 0.02	0.714	0.686 ± 0.016
θ_{path}	-	-0.04 ± 0.04	-	1.30 ± 0.10	-	0.83 ± 0.05

^a Results for these ensembles were discussed in Ref. [28]. They are reshown here for the purpose of comparison.

that required a separate analysis because the second fitting procedure could not be trusted in virtue of its poor performance in modeling the data.

Final estimation for γ is given by $\gamma = \frac{\gamma_1 + \gamma_2}{2} \pm \sqrt{\delta\gamma_{\text{stat}}^2 + \delta\gamma_{\text{syst}}^2}$, where $\delta\gamma_{\text{stat}} = \max(\delta\gamma_1, \delta\gamma_2)$ provides the estimate for the statistical error and $\delta\gamma_{\text{syst}} = \frac{|\gamma_1 - \gamma_2|}{2}$ accounts for the systematic error [28]. With this procedure we have obtained the values of the scaling exponents for the scaling relations (1)-(5) summarized in Table II.

3. Estimating scaling exponents: Distribution functions

Similarly to the scaling exponents for the expectation values of polymer observables, asymptotic values of for the pairs of exponents (θ, t) of Eq. (13) were also obtained through extrapolation to the large-tree limit. More precisely, we followed closely the procedure described in Ref. [35] which combines together the two extrapolation schemes:

1. A fit of the data for θ_ℓ and θ_{tree} to the following 3-parameter fit functions:

$$\begin{aligned} \log \theta_\ell &= a + bN^{-\Delta_0} - b(\Delta - \Delta_0)N^{-\Delta_0} \log N \\ &\equiv a + be^{-\Delta_0 \log N} - b(\Delta - \Delta_0)e^{-\Delta_0 \log N} \log N \end{aligned} \quad (36)$$

and

$$\begin{aligned} \theta_{\text{tree}} &= a + bN^{-\Delta_0} - b(\Delta - \Delta_0)N^{-\Delta_0} \log N \\ &\equiv a + be^{-\Delta_0 \log N} - b(\Delta - \Delta_0)e^{-\Delta_0 \log N} \log N \end{aligned} \quad (37)$$

and analogous expressions for t_ℓ and t_{tree} . Eqs. (36) and (37) correspond to a self-consistent linearisation of the 3 parameter fit $\theta_{\ell, \text{tree}} = a + b\frac{1}{N^\Delta}$ around $\Delta = \Delta_0$. We have carried out a one-dimensional search for the value of Δ_0 for which the fits yield vanishing $N^{-\Delta_0} \log N$ term. Note that we have analyzed data for θ_ℓ (and t_ℓ) in Eq. (36) in the form $\log \theta_\ell$ vs. $\log N$ (log-log), while for θ_{tree} (and t_{tree}) we have used in Eq. (37) data in a log-linear representation, θ_{tree} vs. $\log N$. These two different functional forms have been found to produce the best (statistical significant) fits.

2. In the second method we fixed $\Delta = 1$, and we calculated the corresponding 2-parameter best fits to the same data.

Results from the two fit procedures (including details such as the range of N 's considered for and the statistical significance of the fits) are summarized in Tables VIII and X in the Appendix, as well as their averages (highlighted in boldface) which give our final estimates of scaling exponents. Unfortunately, for θ_{path} and t_{path} an analogous scheme can not be applied due to the limited ranges of ℓ available. In this last cases, our best estimates correspond to simple averages of single values (see boldfaced numbers in Table IX). A summary of the exponents and final errors (calculated as in Sec. III D 2) is given in Table III.

TABLE III. Scaling exponents for distribution functions $p_N(\ell)$ (Eq. (19)), $p_N(\vec{r}|\ell)$ (Eq. (20)) and $p_N(\vec{r})$ (Eq. (21)). For each ensemble, the first and second columns contain the values calculated from Fisher-Pincus (FP) relations, Eqs. (24)-(28), after substitutions of the scaling exponents summarized in Table II: (first column) exponents according to the Flory theory and (second column) exponents estimated from the large-scale behaviors of observables measured from computer simulations. The third column lists the final estimations obtained by extrapolating the scaling exponents evaluated from the fits of the distribution functions.

	2-dimensions								
	Ideal polymer			Good solvent			θ -solvent		
	FP Flory	FP Simulations	Extrapolation Simulations	FP Flory	FP Simulations	Extrapolation Simulations	FP Flory	FP Simulations	Extrapolation Simulations
θ_ℓ	1.00	1.1 ± 0.3	0.96 ± 0.06	0.25	0.35 ± 0.03	0.294 ± 0.001	0.33	0.41 ± 0.03	0.37 ± 0.04
t_ℓ	2.00	1.9 ± 0.2	2.11 ± 0.05	5.00	3.83 ± 0.23	2.48 ± 0.03	4.00	3.46 ± 0.19	2.15 ± 0.07
θ_{path}	0	0	-0.01 ± 0.03	> 0	> 0	1.74 ± 0.09	> 0	> 0	1.2 ± 0.1
t_{path}	2.00	2.04 ± 0.08	2.07 ± 0.06	8.00	6.10 ± 0.33	7.5 ± 0.2	6.00	8.33 ± 2.78	5.8 ± 0.3
θ_{tree}	0.00	-0.30 ± 0.14	0.02 ± 0.005	-0.57	-0.397 ± 0.013	-0.457 ± 0.001	-0.40	-0.36 ± 0.269	-0.379 ± 0.003
t_{tree}	1.33	1.32 ± 0.12	1.34 ± 0.01	3.33	2.66 ± 0.04	2.154 ± 0.002	2.67	2.56 ± 0.66	1.867 ± 0.003

	3-dimensions								
	Ideal polymer ^b			Good solvent ^b			θ -solvent		
	FP Flory	FP Simulations	Extrapolation Simulations	FP Flory	FP Simulations	Extrapolation Simulations	FP Flory	FP Simulations	Extrapolation Simulations
θ_ℓ	1.00	1.04 ± 0.17	1.05 ± 0.2	0.44	0.56 ± 0.05	0.53 ± 0.02	0.57	0.71 ± 0.05	0.68 ± 0.01
t_ℓ	2.00	1.96 ± 0.15	2.02 ± 0.1	3.25	2.78 ± 0.15	2.44 ± 0.01	2.75	2.41 ± 0.10	2.15 ± 0.09
θ_{path}	0	0	0.00	> 0	> 0	1.07 ± 0.08	> 0	> 0	0.53 ± 0.03
t_{path}	2.00	2.04 ± 0.04	2.00	4.50	3.85 ± 0.30	3.8 ± 0.1	3.50	3.18 ± 0.16	3.1 ± 0.1
θ_{tree}	0.00	-0.04 ± 0.04	-0.10 ± 0.2	-1.14	-0.92 ± 0.17	-0.96 ± 0.02	-0.80	-0.53 ± 0.02	-0.57 ± 0.08
t_{tree}	1.33	1.33 ± 0.04	1.26 ± 0.3	2.17	1.92 ± 0.15	2.19 ± 0.02	1.83	1.68 ± 0.01	1.75 ± 0.01

^b Results for these ensembles were discussed in Ref. [35]. They are reshowed here for the purpose of comparison.

IV. RESULTS AND DISCUSSION

A. Average properties

1. Branching statistics

For every polymer ensemble considered in this work, we have computed the average number of branch points or 3-functional nodes of the tree structure, $\langle n_3(N) \rangle$, as a function of polymer size, N . Numerical results for each N are summarized in Table IV and plotted in Fig. 4.

According to Daoud and Joanny [32], for ideal trees the ratio:

$$\frac{\langle n_3(N) \rangle}{\lambda N} \simeq \begin{cases} 1, & \lambda N \gg 1 \\ \lambda N, & \lambda N \ll 1 \end{cases}, \quad (38)$$

where λ defines the asymptotic branching probability per node. As first reported in Ref. [28], the formula by Daoud and Joanny summarizes well our data for ideal trees (see Fig. 4) with branching probability $\lambda \approx 0.4$. Interestingly, it describes well also the frequency of branching points in interacting trees, either self-avoiding trees in good solvent with $\lambda \approx 0.246$ (in $2d$) and $\lambda \approx 0.269$ (in $3d$) or trees in θ -solvent. In particular, these last ones show again $\lambda \approx 0.4$ *i.e.* almost identical to the value of ideal polymers. In spite of evident differences in their spatial structures (see Fig. 1), we conclude that average branching in θ -trees

is mildly affected by volume interactions with respect to their ideal counterparts.

2. Path length statistics

Fig. 5 and Table IV summarize the results for the different observables (see Table I) introduced to quantify how path distances scale in term of the total tree weight N : (a) $\langle L(N) \rangle$: the mean path distance between pairs of nodes; (b) $\langle \delta \ell_{\text{center}}(N) \rangle$: the mean path distance between each node and the tree central node; and (c) $\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$: the mean *longest* path distance of nodes from the tree central node. These three observables are expected to scale with the tree polymer weight as $\sim N^\rho$, see Eq. (2).

Single estimated values for the scaling exponent ρ were derived by applying the procedure described in Sec. III D 2 to each of these quantities, with final results including error bars and the statistical significance being summarized in Table V. Our final best estimates for ρ , obtained by averaging the separate results for $\langle L(N) \rangle$, $\langle \delta \ell_{\text{center}}(N) \rangle$ and $\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$, are given succinctly in Table II and at the top of Fig. 5, while in Table V they are highlighted in boldface with separate annotations for statistical and systematic errors.

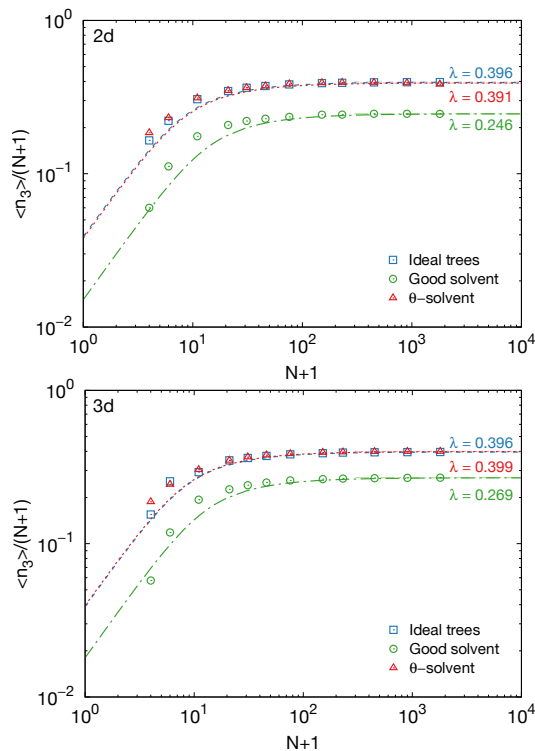


FIG. 4. Average fraction of 3-functional nodes, $\langle n_3(N) \rangle / (N + 1)$, as a function of the total number of tree nodes, $N + 1$. Dashed lines correspond to the analytical expression for ideal trees by Daoud and Joanny, Eq. (38), with corresponding asymptotic branching probabilities λ .

3. Weights of branches vs. path lengths

The scaling exponent ρ describes also the functional relation between the average weight of branches of the trees and the typical path length of the branches.

In order to show this, we analyze the behavior of the average branch weight as a function of the longest path length to the branch root, $\langle N_{\text{br}}(\delta\ell_{\text{root}}^{\text{max}}) \rangle$, and the average branch weight inside a given contour distance from the polymer central node, $\langle N_{\text{center}}(\delta\ell_{\text{center}}) \rangle$. In the limit of large trees (*i.e.*, neglecting finite-chain effects) they should grow as:

$$\langle N_{\text{br}}(\delta\ell_{\text{root}}^{\text{max}}) \rangle = (\delta\ell_{\text{root}}^{\text{max}})^{1/\rho}, \quad (39)$$

$$\langle N_{\text{center}}(\delta\ell_{\text{center}}) \rangle = (\delta\ell_{\text{center}})^{1/\rho}. \quad (40)$$

We have computed $\langle N_{\text{br}}(\delta\ell_{\text{root}}^{\text{max}}) \rangle$ and $\langle N_{\text{center}}(\delta\ell_{\text{center}}) \rangle$ for the different systems studied in this work, see Fig. 6. As expected, in the large- $\delta\ell_{\text{root}}^{\text{max}}$ and large- $\delta\ell_{\text{center}}$ limits they plateau respectively to $\langle N_{\text{br}}(\delta\ell_{\text{root}}^{\text{max}}) \rangle \rightarrow (N - 1)/2$ and $\langle N_{\text{center}}(\delta\ell_{\text{center}}) \rangle \rightarrow N$. At low and intermediate regimes, the scaling laws Eqs. (39) and (40) suggest that one can define a length dependent exponent $\rho = \rho(\delta\ell_{\text{root}}^{\text{max}})$ as:

$$\rho(\delta\ell_{\text{root}}^{\text{max}}) \equiv \left[\frac{\log \langle N_{\text{br}}(\delta\ell_{\text{root}}^{\text{max}} + 1) \rangle - \log \langle N_{\text{br}}(\delta\ell_{\text{root}}^{\text{max}}) \rangle}{\log(\delta\ell_{\text{root}}^{\text{max}} + 1) - \log(\delta\ell_{\text{root}}^{\text{max}})} \right]^{-1}, \quad (41)$$

and an analogous expression for $\rho(\delta\ell_{\text{center}})$. The resulting values of the exponents are shown in the different insets of Fig. 6, they agree well within confidence intervals (shaded green area) with the corresponding final estimates for ρ (see Table II).

4. Branch weight statistics

As anticipated by Eq. (3), the average weight of the polymer branches $\langle N_{\text{br}} \rangle$ scales with the size N in terms of the characteristic exponent ϵ . The asymptotic behaviors for each tree ensemble are reported in Fig. 7, while detailed values for each N are summarized in Table IV. Final estimates for the exponent ϵ are indicated both in the figure and in Table II, while Table V summarizes the statistical details of the fits. We stress that the mathematical relation $\rho = \epsilon$ first pointed out by Janse van Rensburg and Madras [31] for trees in good solvent is accurately verified by all systems studied in this work, and in particular also for θ -trees.

5. Conformational statistics of linear paths

To study the conformational statistics of linear tree paths and determine the corresponding scaling exponent ν_{path} (see Eq. (4)), we have analyzed the end-to-end mean-square spatial distance of paths of average length $\langle L(N) \rangle$, and of average maximal length, $\langle L_{\text{max}}(N) \rangle$. For the scaling relationship Eq. (4), we expect that corresponding mean-square end-to-end spatial distances of these paths obey:

$$\langle R^2(\langle L(N) \rangle) \rangle \sim \langle L(N) \rangle^{2\nu_{\text{path}}}, \quad (42)$$

$$\langle R^2(\langle L_{\text{max}}(N) \rangle) \rangle \sim \langle L_{\text{max}}(N) \rangle^{2\nu_{\text{path}}}. \quad (43)$$

Detailed results for our systems are summarized in Table VI and shown in Fig. 8, together with the final estimates for the exponent ν_{path} (see Table II). The complete statistical details about the fits used to calculate the exponent from the data are summarized in Table VII.

Then, we have measured the mean-square end-to-end distances of linear paths of length ℓ for trees of weight N , $\langle R^2(\ell, N) \rangle$, see the l.h.s panels in Fig. 9. As for quantities $\langle N_{\text{br}}(\delta\ell_{\text{root}}^{\text{max}}) \rangle$ and $\langle N_{\text{center}}(\delta\ell_{\text{center}}) \rangle$ (see Eq. (41)), we define a length-dependent scaling exponent $\nu_{\text{path}}(\ell)$ through the numerical slope in log-log scale:

$$\nu_{\text{path}}(\ell) \equiv \frac{1}{2} \frac{\log \langle R^2(\ell + 1) \rangle - \log \langle R^2(\ell) \rangle}{\log(\ell + 1) - \log(\ell)}. \quad (44)$$

Insets in the l.h.s. plots in Fig. 9 show the values of $\nu_{\text{path}} = \nu_{\text{path}}(\ell)$ for the longest polymer sizes $N \geq 450$, after having been averaged over log-spaced intervals of ℓ for enhancing the visualization. The green shaded region indicates the confidence intervals of the final estimates of ν_{path} summarized in Table II.

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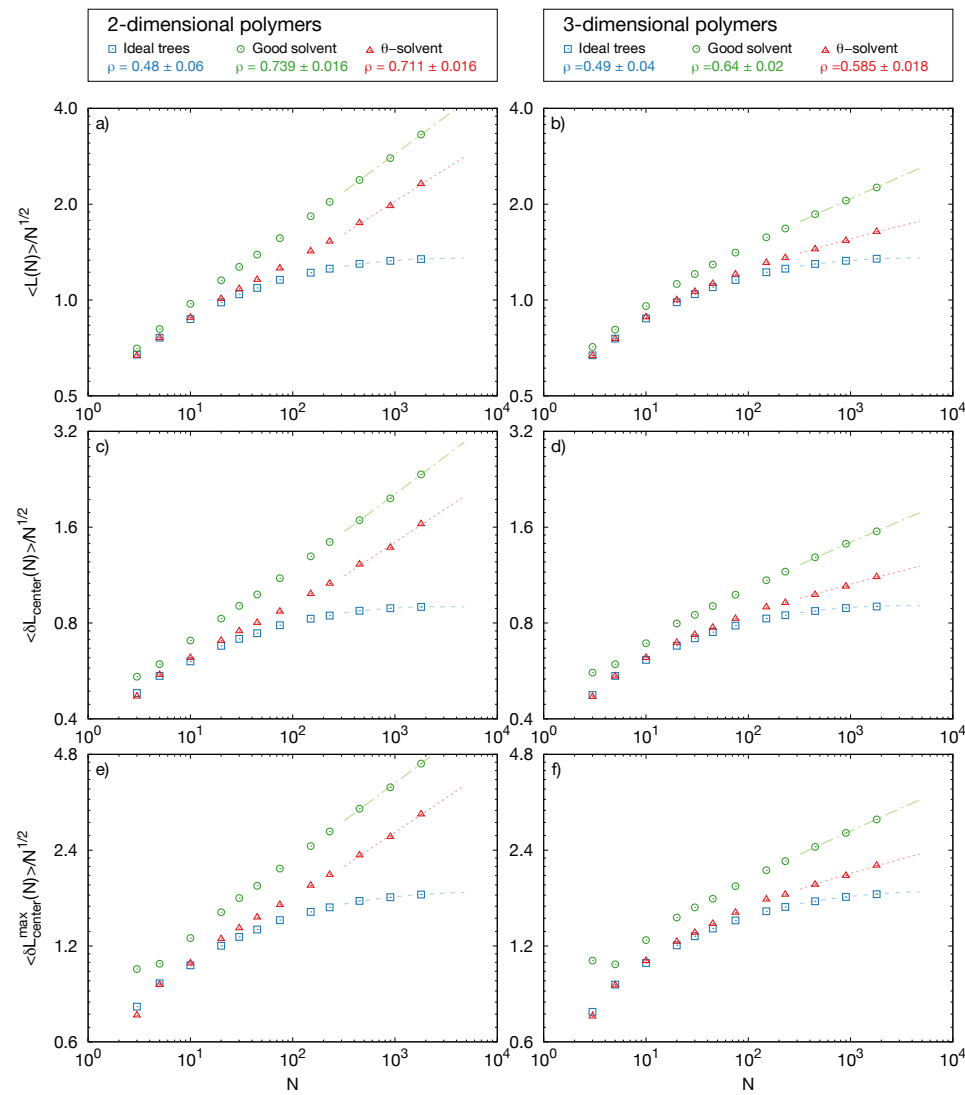


FIG. 5. (a, b) Mean path distance between pairs of nodes, $\langle L(N) \rangle$; (c, d) mean path distance between each node and the tree central node, $\langle \delta \ell_{\text{center}}(N) \rangle$, and (e, f) mean *longest* path distance of nodes from the tree central node, $\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$. Left-hand-side plots correspond to $2d$ polymers, while r.h.s. plots are for $3d$ polymers. Straight dashed line correspond to the large- N behavior $\langle L(N) \rangle \sim \langle \delta \ell_{\text{center}}(N) \rangle \sim \langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle \sim N^\rho$ with scaling exponent ρ . Data for $3d$ ideal polymers and $3d$ self-avoiding polymers in good solvent were already discussed in Ref. [28] and shown here for comparison.

Finally, we conclude this analysis on the spatial conformations of linear paths by discussing the scaling for the average closure probability, $\langle p_c(\ell, N) \rangle$, namely the average number of contacts between pairs of nodes separated by a contour distance ℓ within the tree polymer structure of weight N . Mean-field considerations [20, 28] suggest that $\langle p_c(\ell) \rangle$ should scale as $\sim \langle R^2(\ell) \rangle^{-d/2} \sim \ell^{-d\nu_{\text{path}}}$. Not surprisingly, r.h.s. plots in Fig. 9 show that only ideal polymers obey the mean-field result while interacting polymers display significant deviations which, as anticipated by Eq. (5), can be quantified by introducing the additional scaling exponent θ_{path} , $\langle p_c(\ell) \rangle \sim \ell^{-\nu_{\text{path}}(d+\theta_{\text{path}})}$. Again, we calculated the slope of the data at low and intermediate values of ℓ for trees with $N \geq 450$ and averaged the results over log-spaced intervals to compute

the product $\nu_{\text{path}}\theta_{\text{path}}(\ell)$. Then, we took the results for $N = 1800$ and the values of ν_{path} derived previously from Eqs. (42) and (43) in order to best estimate the corresponding values for θ_{path} . Final results for the different tree ensembles are summarized in Table II.

6. Conformational statistics of lattice trees

Trees spatial conformations were analyzed in terms of the scaling behavior of the expectation value of the square gyration radius with the total tree weight N (see

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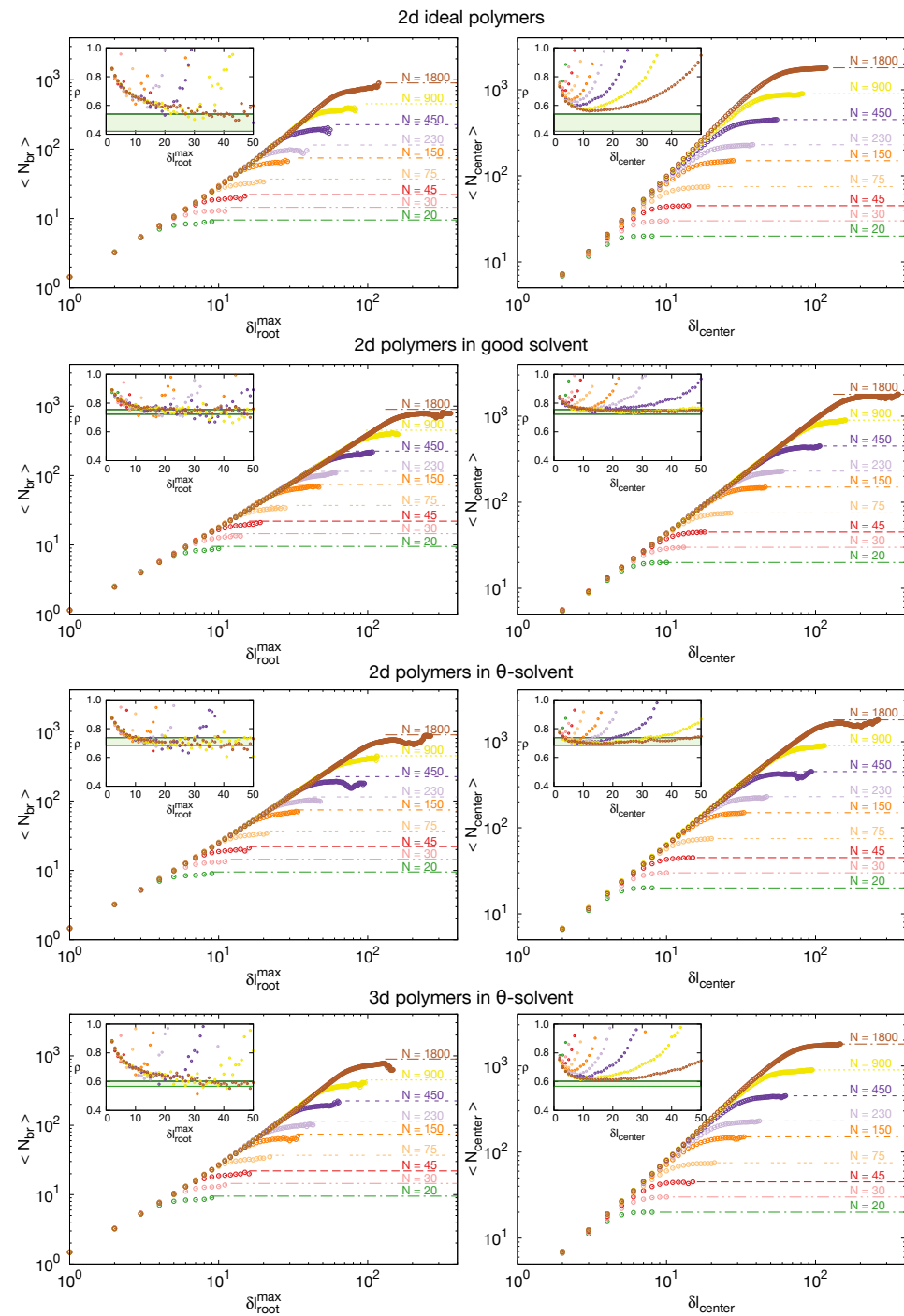


FIG. 6. (Left) Average branch weight as a function of the longest contour distance to the branch root, $\langle N_{br}(\delta l_{root}^{max}) \rangle$. For large δl_{root}^{max} , curves saturate to the corresponding maximal branch weight $(N-1)/2$ (dashed horizontal lines). (Right) Average branch weight, $\langle N_{center}(\delta l_{center}) \rangle$, of paths whose distance from the central node does not exceed δl_{center} . For large δl_{center} , curves saturate to the corresponding total tree weight, N (dashed horizontal lines).

Eq. (1)):

$$\langle R_g^2(N) \rangle \equiv \left\langle \frac{1}{N+1} \sum_{i=1}^{N+1} (\vec{r}_i - \vec{r}_{cm})^2 \right\rangle \sim N^{2\nu}, \quad (45)$$

where \vec{r}_i is the spatial position of the i -th monomer of the tree and $\vec{r}_{cm} \equiv 1/(N+1) \sum_{i=1}^{N+1} \vec{r}_i$ is the spatial position of the tree centre of mass. $\langle R_g^2(N) \rangle$ quantifies polymer swelling in the presence of the solvent [23] with respect to the ideal conditions.

Numerical results for the different ensembles are sum-

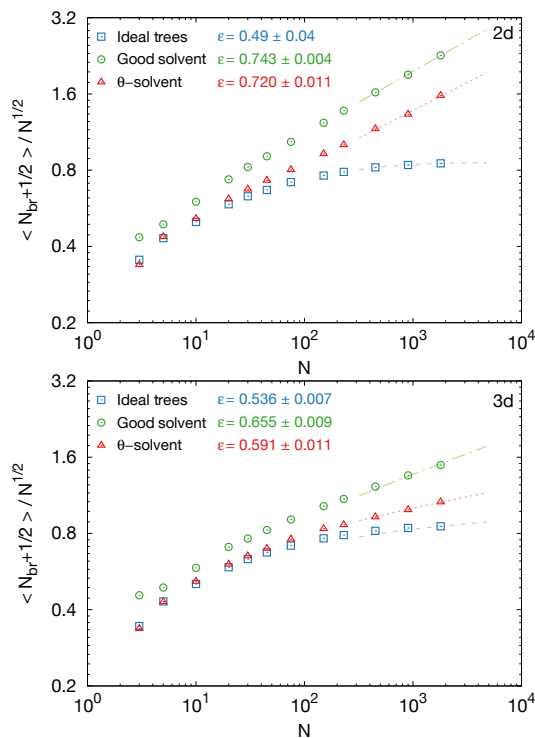


FIG. 7. Average branch weight, $\langle N_{\text{br}}(N) \rangle$ as a function of the total tree mass, N . Straight dashed line correspond to the large- N behavior $\langle N_{\text{br}}(N) \rangle \sim N^\epsilon$ with scaling exponents ϵ . Data for 3d ideal polymers and 3d self-avoiding polymers in good solvent were discussed in previous work [28].

marized in Table VI and shown in Fig. 10. We notice, in particular, that good solvent conditions exhibit the largest swelling while θ -polymers exhibit intermediate behavior between ideal and good solvent conditions due to the balance between repulsive and attractive monomer-monomer spatial interactions, Eq. (32). This observation agrees with the corresponding estimates for the scaling exponent ν whose values and confidence intervals are shown in Fig. 10 and Table II, while the numerical details about their derivation based on best fits to the data are summarized in Table VII. The result for θ -polymers in 2d is, in particular, in good agreement with the accurate value $\nu = 0.5359 \pm 0.0003$ measured by Hsu and Grassberger [56]. It is also worth pointing out that polymer swelling from ideal to θ -solvent conditions as the result of monomer-monomer interactions is mainly affecting the average polymer size while, in comparison, the average internal connectivity appears virtually unperturbed, see Fig. 4.

Finally, we have considered the average polymer shape which can be quantified in terms of the expectation values of the ordered eigenvalues, $\langle \Lambda_1^2(N) \rangle > \langle \Lambda_2^2(N) \rangle > \langle \Lambda_3^2(N) \rangle$ with $\sum_{i=1}^3 \langle \Lambda_i^2(N) \rangle = \langle R_g^2(N) \rangle$, of the 3×3 (in $d = 3$) symmetric polymer gyration tensor, $T = T(N)$,

whose components are given by:

$$T_{\alpha\beta}(N) \equiv \frac{1}{N+1} \sum_{i=1}^{N+1} (\vec{r}_{i,\alpha} - \vec{r}_{\text{cm},\alpha})(\vec{r}_{i,\beta} - \vec{r}_{\text{cm},\beta}), \quad (46)$$

where $\vec{r}_{i,\alpha=x,y,z}$ and $\vec{r}_{\text{cm},\beta=x,y,z}$ stand for the spatial components of the corresponding vectors. Analogous expressions hold for polymers in $d = 2$. Fig. 11 shows the aspect ratios $\langle \Lambda_i^2 \rangle / \langle \Lambda_3^2 \rangle$ for $i = 1, 2$, in particular we may notice that both 2d and 3d ideal polymers appear slightly less a-spherical than their interacting counterparts.

B. Distribution functions

As anticipated in Sec. II B, we conclude this study on the statistical physics properties of lattice trees by discussing the distribution functions for trees connectivity and spatial conformations. In particular, we show that these functions have universal shapes which can be described in terms of the Redner-des Cloizeaux (RdC) theory: the latter implies the existence of new sets of exponents which can be quantitatively related to the exponents describing the scaling of average properties (1)-(5) through generalized Fisher-Pincus (FP) relationships.

1. Path length statistics

We discuss first the distribution functions of linear paths of length ℓ for polymers of size N , $p_N(\ell)$. As shown in Fig. 12, this function obeys the universal RdC scaling form described by Eq. (19). We calculated then the pair of exponents (θ_ℓ, t_ℓ) for each polymer weight of N by fitting the plotted data with rescaled path length $x \equiv \ell / \langle L(N) \rangle$ to Eq. (13) (see Table VIII for details). Then, we obtained the final estimation for the pair (θ_ℓ, t_ℓ) by extrapolating the finite- N results using Eq. (36) and the related methods explained in Sec. III D 3: the reconstructed distribution functions are shown for comparison as black curves in Fig. 12.

Finally, we compared the extrapolated values for (θ_ℓ, t_ℓ) to the quantitative Fisher-Pincus (FP) expressions (Eqs. (24) and (25)) relating those exponents to ρ . This task has been summarized in Table III showing the final estimations for (θ_ℓ, t_ℓ) and the results from the FP formulas with ρ being equal to the Flory values and to the final values estimated from the present simulations.

Although we may appreciate the power of the Flory theory which, once again, in spite of its simplicity seems to agree well with the data in a close-to-quantitative manner, we may also notice that, in all non-ideal cases, it underestimates systematically the exponent θ_ℓ and overestimates t_ℓ , with differences between theory and simulations more pronounced in $d = 2$. Finally, the exponents (θ_ℓ, t_ℓ) agree well with the predictions of the FP relations with ρ corresponding to the values estimated in this work.

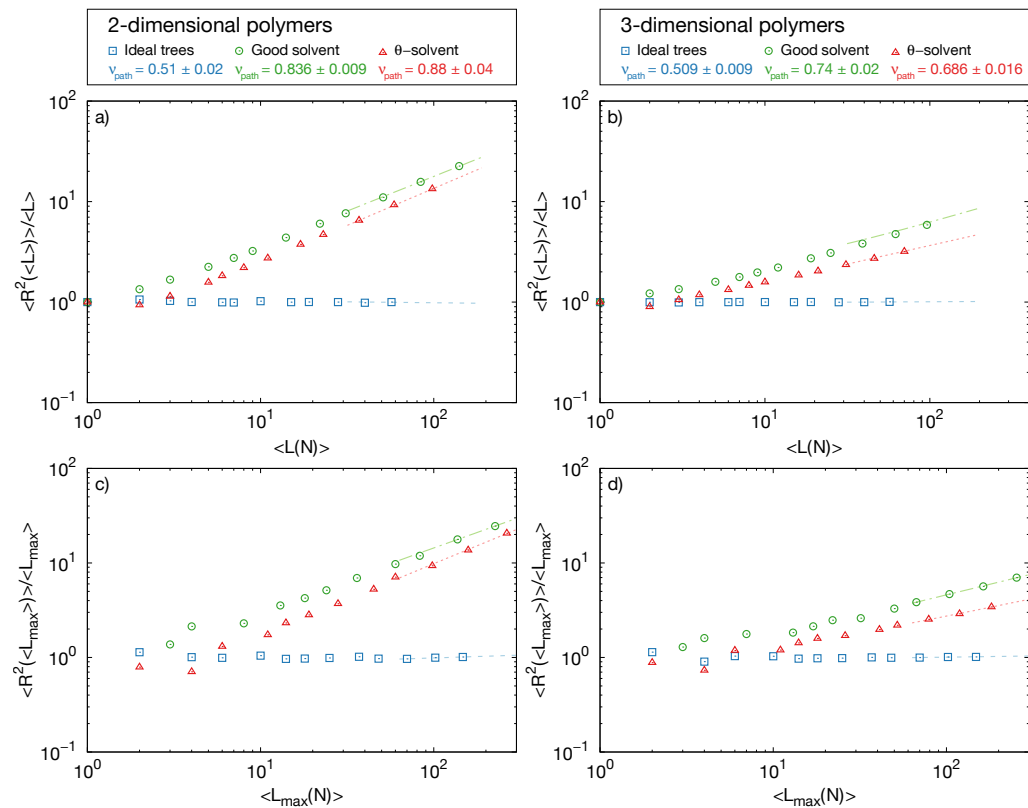


FIG. 8. (a,b) Mean square end-to-end spatial distance, $\langle R^2(\ell) \rangle$, of paths of length $\ell = \langle L(N) \rangle$; (c, d) mean square end-to-end spatial distance, $\langle R^2(L_{\max}) \rangle$, of the longest paths. L.h.s. plots correspond to 2d polymers, while r.h.s. plots are for 3d polymers. Straight dashed lines correspond to the large- N scaling behavior, Eqs. (42) and (43), with exponent ν_{path} . Data for 3d ideal polymers and 3d self-avoiding polymers in good solvent were discussed in previous work [28].

2. Conformational statistics of linear paths

We considered the distribution function $p_N(\vec{r}|\ell)$ of end-to-end vectors of linear paths of length ℓ on trees of weight N and we plot them as a function of the scaling variable $x \equiv |\vec{r}|/\sqrt{\langle R^2(\ell, N) \rangle}$. As shown in Fig. 13 the data from different N 's collapse to a single universal shape, which obeys the RdC functional form (13) with given exponents θ_{path} and t_{path} . In the ideal 2d case, $p_N(\vec{r}|\ell)$ is well described by the Gaussian function with $\theta_{\text{path}} = 2$ and $t_{\text{path}} = 0$ (black curve in Fig. 13(a)). In the other cases, the exponents θ_{path} and t_{path} have been found by best fits of the RdC expression to the data for different N 's and ℓ 's (see Table IX). As briefly mentioned in Sec. III D 3, our final estimations for the pairs $(\theta_{\text{path}}, t_{\text{path}})$ were given by averaging the single values determined for trees with $N \geq 450$ and path lengths $\ell = 16, 32, 64$, the corresponding RdC functions shown as black curves in panels (b)-(d) of Fig. 13. In fact, no extrapolation was attempted in this case due to the limited range of path lengths available for our systems.

Finally, in Table III we compare the estimated values for t_{path} to the Fisher-Pincus (FP) relationship from Eq. (26) upon substitution of theoretical Flory values and our numerical results. The agreement is overall good,

with values from the Flory theory typically overestimating the numerical predictions. It was remarked [35] that $\theta_{\text{path}} > 0$, otherwise there seems to be no relation between this exponent and the others: this suggests that θ_{path} should be regarded as a genuinely novel exponent.

3. Conformational statistics of lattice trees

Finally, we conclude this analysis by considering the distribution functions for the end-to-end distances between pairs of nodes of trees of weight N , $p_N(\vec{r})$, as a function of the rescaled distance $x \equiv |\vec{r}|/\sqrt{2\langle R_g^2(N) \rangle}$. As in the previous cases, the curves from different polymers superimpose (see Fig. 14) and agree with the Redner-de Cloizeaux functional form Eq. (13). The corresponding pairs of exponents $(\theta_{\text{tree}}, t_{\text{tree}})$ were calculated by first fitting data to (13) for every N (see Table X) and then extrapolating the results according to the numerical procedure outlined in Sec. III D 3. Final results are shown in Table III and compared to the predictions of Fisher-Pincus relations Eqs. (27) and (28) upon substitutions of the theoretical (Flory) and numerical values for θ_{path} and ν .

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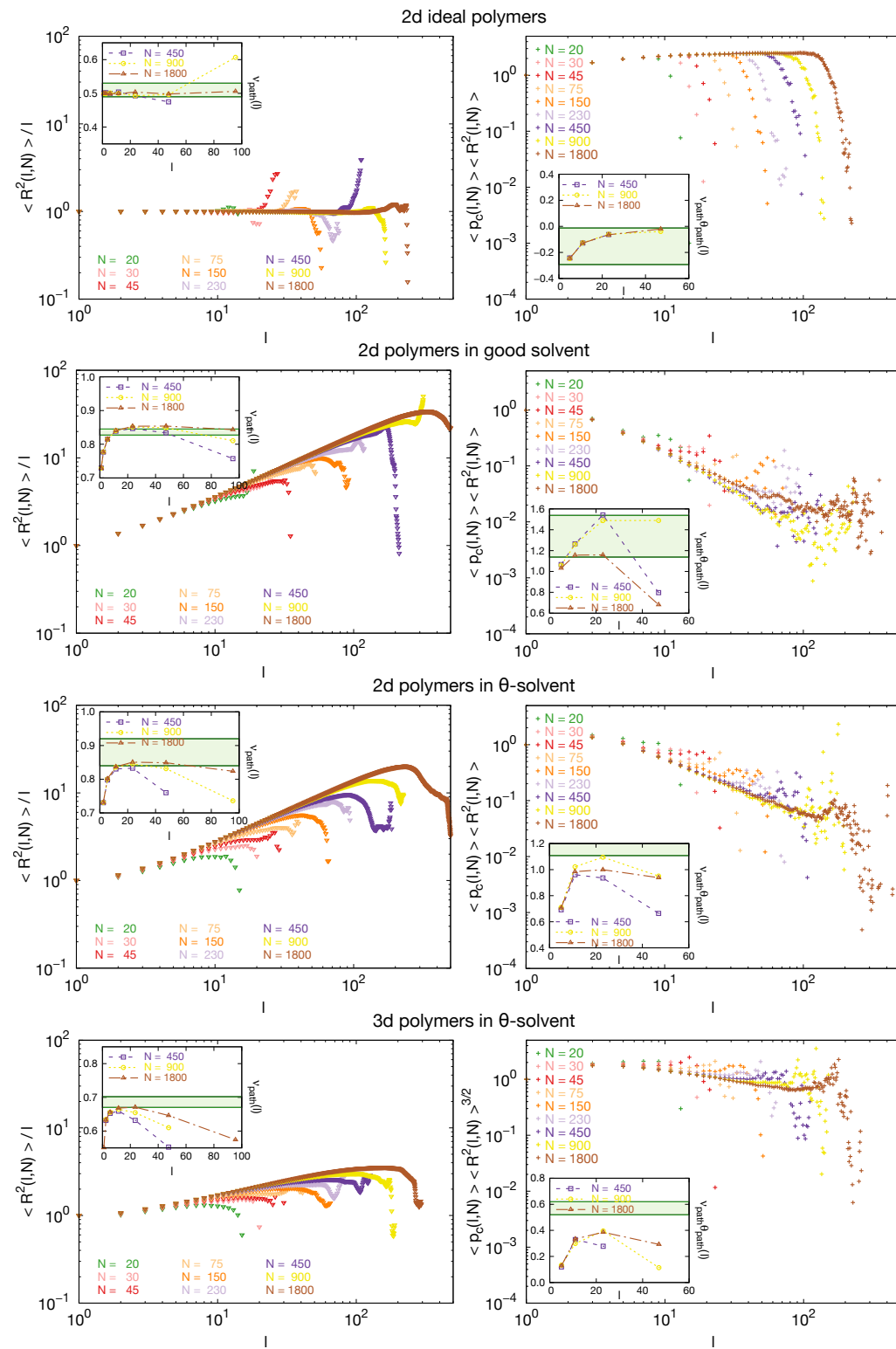


FIG. 9. (Left) Mean-square end-to-end distance, $\langle R^2(\ell, N) \rangle$, of linear paths of length ℓ . Insets: scaling exponent $\langle \nu_{\text{path}}(\ell) \rangle$, averaged over log-spaced intervals for polymer size $N \geq 450$. (Right) Mean closure probabilities, $\langle p_c(\ell, N) \rangle$, between ends of linear paths of length ℓ normalized to the mean-field expectation value $\langle R^2(\ell, N) \rangle^{d/2}$. Insets: scaling exponent $\langle \nu_{\text{path}} \theta_{\text{path}}(\ell) \rangle$, averaged over log-spaced intervals for polymer size $N \geq 450$.

V. CONCLUSIONS

Following the outline of recent work [20, 28, 35] by our group, in the present paper we have presented a sys-

tematic, quantitative analysis regarding the scaling prop-

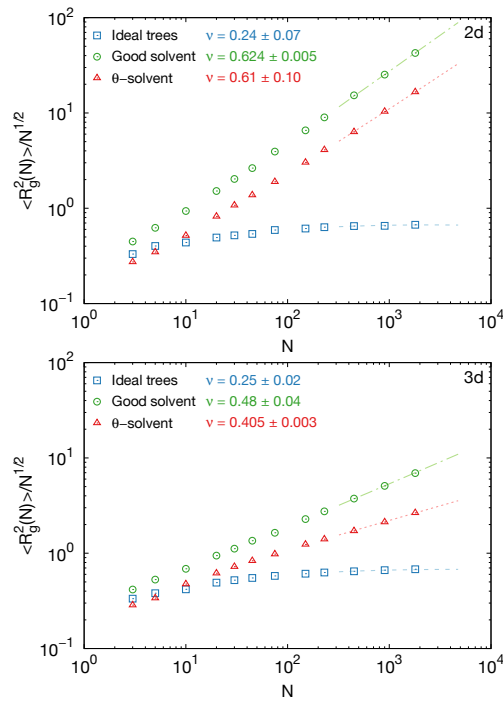


FIG. 10. Mean square gyration radius, $\langle R_g^2(N) \rangle$. Straight lines correspond to the large- N behavior $\langle R_g^2(N) \rangle \sim N^{2\nu}$ with scaling exponent ν . Data for 3d ideal polymers and 3d self-avoiding polymers in good solvent were discussed in previous work [28].

erties of single conformations of branching polymers in θ -solvent conditions and annealed connectivity. To this purpose, we have employed a mix of theoretical considerations based on the Flory theory, results of on-lattice Monte Carlo computer simulations and rigorous numerical extrapolation methods to derive best estimates for scaling exponents of average chain properties and distribution functions. We highlight three main crucial aspects of the present article.

First, the need to generalize the Monte Carlo method first discussed in Ref. [20, 28] in order to model properly the effects of the θ -solvent. This task (Sec. III C) has been accomplished by suitably tailoring the force-field in order to recover the most accurate value to date [53] of the scaling exponent ν relating the average polymer size and the polymer weight N in 3d, see Eq. (1). Remarkably, we found that the same interaction parameters found for the 3d case appear appropriate also for 2d polymers (Sec. IV A 6) without additional fine calibration: this shows the generality and robustness of our model.

Second, we performed a detailed analysis of the average properties, Eqs. (1)-(5), of tree polymers focusing, in particular, on the accurate derivation of the corresponding scaling exponents. This analysis, interesting *per se*, has been carried out by systematically comparing our data to the predictions of the Flory theory. The study, which confirms and completes the discussion first started

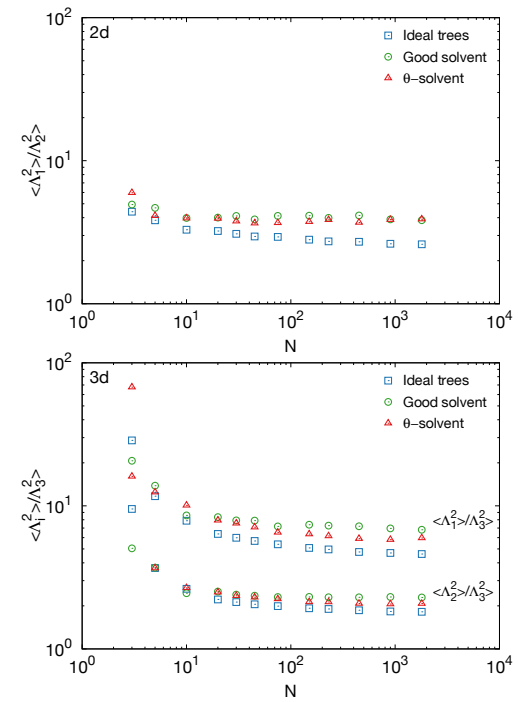


FIG. 11. Average polymer aspect ratios from the expectation values of the gyration tensor eigenvalues, $\langle \Lambda_1^2 \rangle > \langle \Lambda_2^2 \rangle > \langle \Lambda_3^2 \rangle$. Data for 3d ideal polymers and 3d self-avoiding polymers in good solvent were discussed in previous work [28].

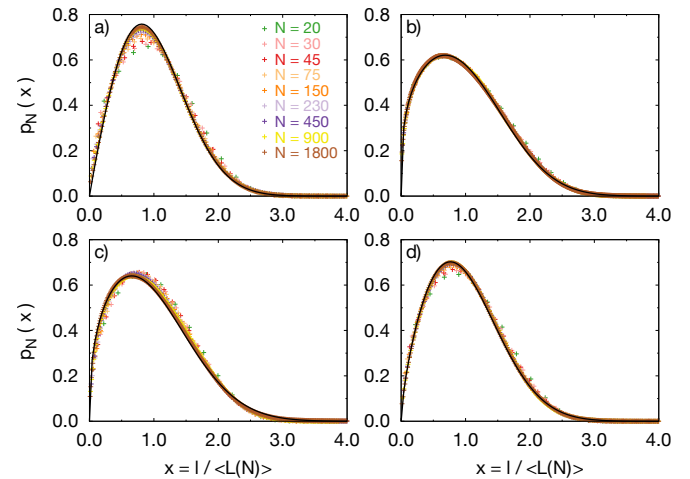


FIG. 12. Distribution functions $p_N(\ell)$ of linear paths of length ℓ for polymers of size N : (a) 2d ideal polymers, (b) 2d polymers in good solvent, (c) 2d polymers in θ -solvent and (d) 3d polymers in θ -solvent. Black solid lines correspond to the Redner-des Cloizeaux function Eq. (13) computed with the final parameters (θ_ℓ, t_ℓ) summarized in Table III.

in Refs. [20, 28], supports the idea that simple Flory theories [34, 37, 38] can indeed be used to gain insight into the physics of polymers with branched architectures, in particular in rationalizing the trends of critical quantities with respect to the typical tree path length or polymer mass (see Table II).

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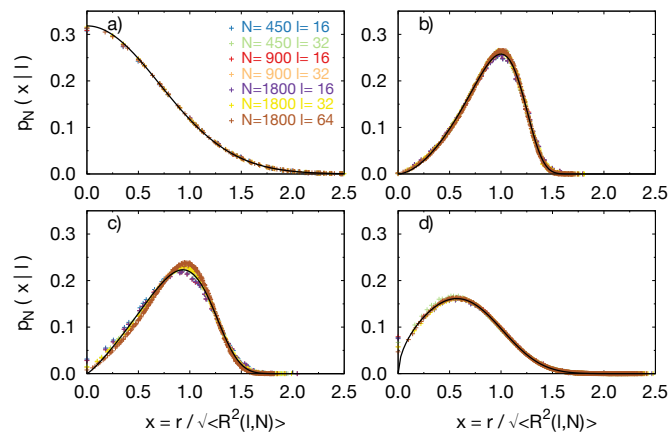


FIG. 13. Distribution functions $p_N(\vec{r}|\ell)$ of end-to-end spatial vectors \vec{r} for linear paths of length ℓ on polymers of size N : (a) $2d$ ideal polymers, (b) $2d$ polymers in good solvent, (c) $2d$ polymers in θ -solvent and (d) $3d$ polymers in θ -solvent. Black solid lines correspond to the Redner-des Cloizeaux function Eq. (13) computed by using the final estimations of parameters $(\theta_{\text{path}}, t_{\text{path}})$ summarized in Table III.

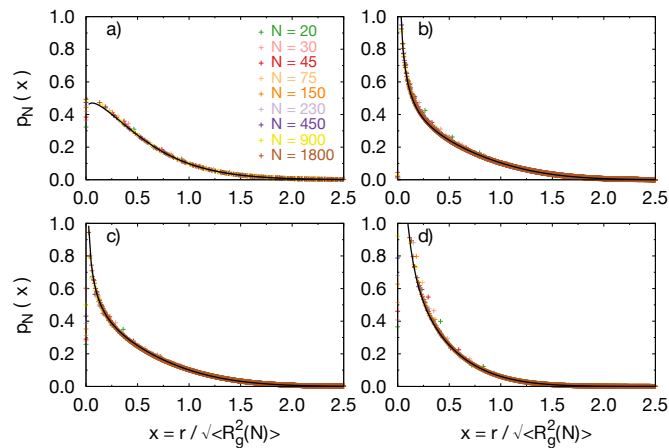


FIG. 14. Distribution functions $p_N(\vec{r})$ of end-to-end spatial vectors \vec{r} between pairs of nodes on tree polymers of size N : (a) $2d$ ideal polymers, (b) $2d$ polymers in good solvent, (c) $2d$ polymers in θ -solvent and (d) $3d$ polymers in θ -solvent. Black solid lines correspond to the Redner-des Cloizeaux function Eq. (13) computed by using the final estimations of parameters $(\theta_{\text{tree}}, t_{\text{tree}})$ summarized in Table III.

Third, by measuring and discussing the complete statistics of path lengths and spatial polymer conformations we were able to move beyond the Flory approximation and, thus, fill the remaining gaps. In particular, we confirmed that the distribution functions obey the Redner-des Cloizeaux statistics (Eq. (13)) with novel exponents which can be quantitatively understood by suitably generalizing the classical Fisher-Pincus theory of polymer physics (see Table III).

Taken together, the results of our work draw a comprehensive picture of the static properties of polymers with annealed branching architectures in θ -solvent conditions.

Nonetheless, the interest of this piece of research lies not only in the numerical results described, which allow to compare different models and theories regarding branching polymers. As nowadays a large variety of branching polymers can be easily synthesized [57], our results, together with previous ones [20, 28, 34, 35] published in this topic, may also serve as a reference guide for understanding the physical properties of branched polymers in many practical contexts.

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APPENDIX

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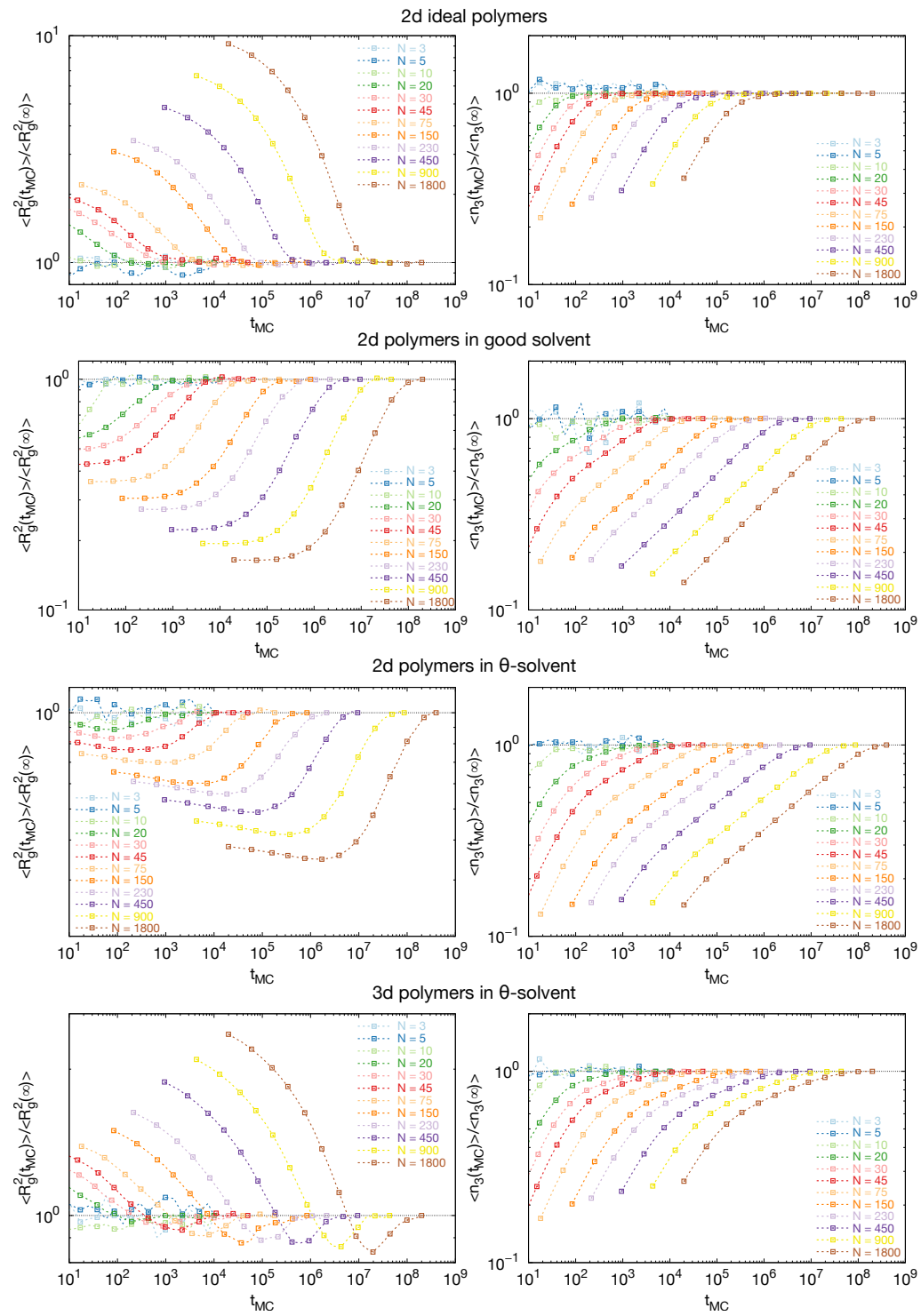


FIG. 15. Monitoring the equilibration of $2d$ and $3d$ lattice trees of total weight N in terms of Monte Carlo time steps (t_{MC}) of the “amoeba” algorithm (Sec. III B 1). (Left) Mean-square gyration radius. (Right) Mean number of branching nodes. Both quantities are normalized to the corresponding asymptotic values.

TABLE IV. Path length statistics and branching statistics I. Expectation values and confidence intervals of corresponding observables for tree polymers of total weight N .

2d ideal polymers					
N	$\langle L(N) \rangle$	$\langle \delta \ell_{\text{center}}(N) \rangle$	$\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$	$\langle N_{\text{br}}(N) \rangle$	$\langle n_3 \rangle$
3	1.168 ± 0.006	0.835 ± 0.012	1.340 ± 0.047	0.613 ± 0.072	0.660 ± 0.048
5	1.703 ± 0.009	1.220 ± 0.009	2.050 ± 0.022	0.964 ± 0.072	1.330 ± 0.057
10	2.754 ± 0.016	1.917 ± 0.019	3.300 ± 0.046	1.581 ± 0.073	3.370 ± 0.063
20	4.395 ± 0.009	3.033 ± 0.010	5.375 ± 0.022	2.624 ± 0.025	7.278 ± 0.029
30	5.708 ± 0.014	3.909 ± 0.014	7.019 ± 0.030	3.457 ± 0.026	11.273 ± 0.035
45	7.321 ± 0.019	4.986 ± 0.019	9.072 ± 0.037	4.483 ± 0.029	17.188 ± 0.044
75	10.025 ± 0.030	6.820 ± 0.029	12.535 ± 0.057	6.219 ± 0.036	29.071 ± 0.056
150	14.923 ± 0.050	10.109 ± 0.046	18.810 ± 0.089	9.332 ± 0.051	58.876 ± 0.075
230	19.022 ± 0.067	12.804 ± 0.061	24.075 ± 0.113	11.947 ± 0.064	90.546 ± 0.096
450	27.565 ± 0.105	18.545 ± 0.095	35.274 ± 0.174	17.417 ± 0.095	177.864 ± 0.140
900	39.822 ± 0.153	26.709 ± 0.138	51.232 ± 0.246	25.165 ± 0.133	356.388 ± 0.199
1800	57.048 ± 0.156	38.115 ± 0.137	73.877 ± 0.246	36.109 ± 0.134	713.441 ± 0.189

2d polymers in good solvent					
N	$\langle L(N) \rangle$	$\langle \delta \ell_{\text{center}}(N) \rangle$	$\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$	$\langle N_{\text{br}}(N) \rangle$	$\langle n_3 \rangle$
3	1.220 ± 0.005	0.940 ± 0.011	1.760 ± 0.043	0.753 ± 0.072	0.240 ± 0.043
5	1.813 ± 0.010	1.328 ± 0.014	2.360 ± 0.048	1.094 ± 0.073	0.670 ± 0.053
10	3.076 ± 0.022	2.228 ± 0.024	4.020 ± 0.051	1.899 ± 0.075	1.930 ± 0.081
20	5.158 ± 0.014	3.692 ± 0.014	6.855 ± 0.029	3.291 ± 0.027	4.365 ± 0.035
30	6.960 ± 0.021	4.961 ± 0.020	9.299 ± 0.041	4.508 ± 0.031	6.843 ± 0.046
45	9.317 ± 0.029	6.595 ± 0.028	12.441 ± 0.054	6.090 ± 0.036	10.513 ± 0.054
75	13.545 ± 0.042	9.582 ± 0.040	18.187 ± 0.076	8.967 ± 0.046	17.843 ± 0.072
150	22.446 ± 0.073	15.867 ± 0.067	30.285 ± 0.132	15.073 ± 0.072	36.685 ± 0.097
230	30.828 ± 0.102	21.795 ± 0.093	41.661 ± 0.177	20.820 ± 0.096	56.169 ± 0.117
450	50.551 ± 0.160	35.703 ± 0.146	68.730 ± 0.284	34.426 ± 0.149	111.058 ± 0.170
900	83.736 ± 0.277	59.099 ± 0.258	113.453 ± 0.489	57.152 ± 0.258	221.690 ± 0.241
1800	140.430 ± 0.364	99.362 ± 0.336	190.439 ± 0.596	96.369 ± 0.331	443.047 ± 0.279

2d polymers in θ -solvent					
N	$\langle L(N) \rangle$	$\langle \delta \ell_{\text{center}}(N) \rangle$	$\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$	$\langle N_{\text{br}}(N) \rangle$	$\langle n_3 \rangle$
3	1.158 ± 0.005	0.815 ± 0.011	1.260 ± 0.044	0.587 ± 0.072	0.740 ± 0.044
5	1.700 ± 0.008	1.230 ± 0.009	2.030 ± 0.017	0.976 ± 0.072	1.390 ± 0.055
10	2.784 ± 0.013	1.968 ± 0.016	3.350 ± 0.048	1.628 ± 0.072	3.430 ± 0.056
20	4.512 ± 0.010	3.150 ± 0.010	5.649 ± 0.022	2.755 ± 0.025	7.316 ± 0.030
30	5.934 ± 0.015	4.134 ± 0.015	7.484 ± 0.032	3.689 ± 0.027	11.296 ± 0.036
45	7.774 ± 0.022	5.379 ± 0.021	9.905 ± 0.044	4.889 ± 0.031	17.206 ± 0.043
75	10.891 ± 0.032	7.534 ± 0.030	14.005 ± 0.060	6.957 ± 0.038	29.286 ± 0.053
150	17.429 ± 0.056	12.100 ± 0.052	22.771 ± 0.099	11.374 ± 0.057	59.142 ± 0.079
230	23.178 ± 0.082	16.150 ± 0.075	30.468 ± 0.139	15.274 ± 0.078	90.834 ± 0.101
450	37.039 ± 0.149	25.939 ± 0.137	49.093 ± 0.239	24.713 ± 0.134	177.483 ± 0.166
900	59.240 ± 0.202	41.435 ± 0.184	79.319 ± 0.347	39.894 ± 0.185	352.540 ± 0.404
1800	98.262 ± 0.323	69.474 ± 0.307	132.010 ± 0.477	66.873 ± 0.284	690.381 ± 0.808

3d polymers in θ -solvent					
N	$\langle L(N) \rangle$	$\langle \delta \ell_{\text{center}}(N) \rangle$	$\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$	$\langle N_{\text{br}}(N) \rangle$	$\langle n_3 \rangle$
3	1.156 ± 0.005	0.813 ± 0.011	1.250 ± 0.043	0.583 ± 0.072	0.750 ± 0.044
5	1.688 ± 0.008	1.217 ± 0.008	2.020 ± 0.014	0.960 ± 0.071	1.460 ± 0.054
10	2.794 ± 0.015	1.968 ± 0.017	3.410 ± 0.049	1.635 ± 0.073	3.340 ± 0.065
20	4.470 ± 0.010	3.105 ± 0.010	5.540 ± 0.023	2.704 ± 0.025	7.340 ± 0.029
30	5.820 ± 0.015	4.024 ± 0.015	7.241 ± 0.032	3.570 ± 0.027	11.400 ± 0.035
45	7.542 ± 0.021	5.193 ± 0.020	9.457 ± 0.041	4.678 ± 0.030	17.421 ± 0.042
75	10.429 ± 0.032	7.152 ± 0.031	13.236 ± 0.060	6.568 ± 0.038	29.327 ± 0.056
150	16.038 ± 0.055	10.988 ± 0.051	20.600 ± 0.097	10.228 ± 0.055	59.350 ± 0.079
230	20.563 ± 0.073	14.041 ± 0.067	26.423 ± 0.126	13.130 ± 0.069	91.420 ± 0.096
450	30.650 ± 0.109	20.815 ± 0.100	39.702 ± 0.182	19.712 ± 0.100	179.529 ± 0.138
900	46.057 ± 0.173	31.272 ± 0.156	59.867 ± 0.280	29.745 ± 0.155	359.534 ± 0.188
1800	69.520 ± 0.187	47.381 ± 0.167	91.185 ± 0.301	45.169 ± 0.164	719.044 ± 0.190

TABLE V. Path length statistics and branching statistics II. Values for the critical exponents ρ and ϵ , obtained from best fits of model functions with ($\Delta > 0$) and without ($\Delta = 0$) correction-to-scaling term (see Sec. III D 2, for details) to the corresponding numerical data reported in Table IV. Final estimates with systematic and statistical errors are highlighted in boldface.

2d ideal polymers				
Observable	$\langle L(N) \rangle$	$\langle \delta \ell_{\text{center}}(N) \rangle$	$\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$	$\langle N_{\text{br}}(N) \rangle$
Δ	0.262 ± 0.032	0.205 ± 0.055	0.266 ± 0.045	0.309 ± 0.048
DOF	6	6	6	6
$\tilde{\chi}^2$	0.532	0.618	0.453	0.267
\mathcal{Q}	0.784	0.716	0.844	0.953
Exponent	$\rho = 0.452 \pm 0.015$	$\rho = 0.423 \pm 0.038$	$\rho = 0.458 \pm 0.021$	$\epsilon = 0.454 \pm 0.019$
Δ	0	0	0	0
DOF	1	1	1	1
$\tilde{\chi}^2$	0.868	0.581	0.404	0.313
\mathcal{Q}	0.351	0.446	0.525	0.576
Exponent	$\rho = 0.524 \pm 0.002$	$\rho = 0.519 \pm 0.002$	$\rho = 0.533 \pm 0.002$	$\epsilon = 0.525 \pm 0.002$
Average	$\rho = 0.485 \pm 0.042 \pm 0.038$			$\epsilon = 0.490 \pm 0.036 \pm 0.019$
2d polymers in good solvent				
Observable	$\langle L(N) \rangle$	$\langle \delta \ell_{\text{center}}(N) \rangle$	$\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$	$\langle N_{\text{br}}(N) \rangle$
Δ	0.908 ± 0.785	0.383 ± 0.412	1.094 ± 8.399	0.770 ± 0.523
DOF	4	5	5	6
$\tilde{\chi}^2$	0.776	0.835	1.113	0.567
\mathcal{Q}	0.541	0.525	0.351	0.757
Exponent	$\rho = 0.738 \pm 0.003$	$\rho = 0.747 \pm 0.016$	$\rho = 0.739 \pm 0.003$	$\epsilon = 0.743 \pm 0.004$
Δ	0	0	0	0
DOF	1	1	1	1
$\tilde{\chi}^2$	2.525	2.325	2.771	2.166
\mathcal{Q}	0.112	0.127	0.096	0.141
Exponent	$\rho = 0.738 \pm 0.003$	$\rho = 0.739 \pm 0.004$	$\rho = 0.736 \pm 0.004$	$\epsilon = 0.743 \pm 0.004$
Average	$\rho = 0.739 \pm 0.004 \pm 0.016$			$\epsilon = 0.743 \pm 0.000 \pm 0.004$
2d polymers in θ -solvent				
Observable	$\langle L(N) \rangle$	$\langle \delta \ell_{\text{center}}(N) \rangle$	$\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$	$\langle N_{\text{br}}(N) \rangle$
Δ	—	—	—	—
DOF	—	—	—	—
$\tilde{\chi}^2$	—	—	—	—
\mathcal{Q}	—	—	—	—
Exponent	—	—	—	—
Δ	0	0	0	0
DOF	1	1	1	1
$\tilde{\chi}^2$	18.065	18.679	7.731	10.616
\mathcal{Q}	$2 \cdot 10^{-5}$	$1 \cdot 10^{-5}$	0.005	0.001
Exponent	$\rho = 0.706 \pm 0.011$	$\rho = 0.713 \pm 0.015$	$\rho = 0.716 \pm 0.008$	$\epsilon = 0.720 \pm 0.011$
Average	$\rho = 0.711 \pm 0.004 \pm 0.015$			$\epsilon = 0.720 \pm 0.000 \pm 0.011$
3d polymers in θ -solvent				
Observable	$\langle L(N) \rangle$	$\langle \delta \ell_{\text{center}}(N) \rangle$	$\langle \delta \ell_{\text{center}}^{\text{max}}(N) \rangle$	$\langle N_{\text{br}}(N) \rangle$
Δ	0.473 ± 0.088	0.506 ± 0.163	0.463 ± 0.154	0.667 ± 0.150
DOF	6	6	6	6
$\tilde{\chi}^2$	1.503	1.672	2.056	1.081
\mathcal{Q}	0.173	0.123	0.055	0.371
Exponent	$\rho = 0.572 \pm 0.008$	$\rho = 0.576 \pm 0.010$	$\rho = 0.577 \pm 0.015$	$\epsilon = 0.584 \pm 0.008$
Δ	0	0	0	0
DOF	1	1	1	1
$\tilde{\chi}^2$	0.263	0.525	0.840	0.269
\mathcal{Q}	0.608	0.469	0.359	0.604
Exponent	$\rho = 0.591 \pm 0.001$	$\rho = 0.594 \pm 0.002$	$\rho = 0.600 \pm 0.003$	$\epsilon = 0.599 \pm 0.002$
Average	$\rho = 0.585 \pm 0.011 \pm 0.015$			$\epsilon = 0.591 \pm 0.007 \pm 0.008$

TABLE VI. Conformational statistics of lattice trees I. Expectation values and confidence intervals of corresponding observables for tree polymers of total weight N .

2d ideal polymers				
N	$\langle R_g^2(N) \rangle$	$\langle R^2(\langle L \rangle) \rangle$	L_{\max}	$\langle R^2(L_{\max}) \rangle$
3	0.573 ± 0.019	1.000 ± 0.000	2.340 ± 0.047	2.273 ± 0.149
5	0.899 ± 0.031	2.122 ± 0.057	3.670 ± 0.057	4.030 ± 0.310
10	1.382 ± 0.054	3.085 ± 0.104	6.130 ± 0.081	5.945 ± 0.537
20	2.207 ± 0.029	4.009 ± 0.038	10.260 ± 0.042	10.448 ± 0.282
30	2.847 ± 0.039	5.972 ± 0.069	13.571 ± 0.057	13.538 ± 0.385
45	3.610 ± 0.049	6.936 ± 0.074	17.647 ± 0.074	17.532 ± 0.500
75	5.117 ± 0.074	10.205 ± 0.116	24.550 ± 0.114	24.690 ± 0.760
150	7.501 ± 0.105	14.995 ± 0.166	37.120 ± 0.178	37.636 ± 1.133
230	9.599 ± 0.134	19.028 ± 0.212	47.666 ± 0.226	46.640 ± 1.452
450	13.794 ± 0.195	28.014 ± 0.325	70.060 ± 0.347	67.656 ± 2.158
900	19.631 ± 0.256	39.384 ± 0.433	101.976 ± 0.492	101.510 ± 3.351
1800	28.470 ± 0.286	57.007 ± 0.470	147.275 ± 0.491	148.828 ± 3.511

2d polymers in good solvent				
N	$\langle R_g^2(N) \rangle$	$\langle R^2(\langle L \rangle) \rangle$	L_{\max}	$\langle R^2(L_{\max}) \rangle$
3	0.776 ± 0.022	1.000 ± 0.000	2.760 ± 0.043	4.120 ± 0.226
5	1.394 ± 0.035	2.687 ± 0.036	4.330 ± 0.053	8.530 ± 0.408
10	2.957 ± 0.083	5.019 ± 0.070	7.590 ± 0.094	18.370 ± 0.966
20	6.784 ± 0.060	11.207 ± 0.050	13.197 ± 0.055	46.154 ± 0.822
30	11.118 ± 0.101	19.262 ± 0.091	18.098 ± 0.080	76.408 ± 1.457
45	17.695 ± 0.156	29.027 ± 0.124	24.377 ± 0.107	123.440 ± 2.318
75	34.080 ± 0.304	61.383 ± 0.290	35.888 ± 0.151	249.581 ± 4.470
150	80.480 ± 0.697	132.646 ± 0.595	60.072 ± 0.263	583.060 ± 11.008
230	136.353 ± 1.232	237.233 ± 1.066	82.826 ± 0.355	987.330 ± 18.925
450	324.685 ± 2.982	561.806 ± 2.570	136.958 ± 0.566	2429.080 ± 45.923
900	757.177 ± 6.516	1319.930 ± 5.515	226.413 ± 0.977	5544.630 ± 108.576
1800	1807.139 ± 10.327	3156.800 ± 9.886	380.355 ± 1.192	13151.400 ± 171.613

2d polymers in θ -solvent				
N	$\langle R_g^2(N) \rangle$	$\langle R^2(\langle L \rangle) \rangle$	L_{\max}	$\langle R^2(L_{\max}) \rangle$
3	0.474 ± 0.021	1.000 ± 0.000	2.260 ± 0.044	1.580 ± 0.149
5	0.776 ± 0.027	1.873 ± 0.059	3.610 ± 0.055	2.820 ± 0.202
10	1.635 ± 0.053	3.418 ± 0.084	6.280 ± 0.062	7.847 ± 0.562
20	3.672 ± 0.037	7.867 ± 0.056	10.787 ± 0.043	19.076 ± 0.438
30	5.887 ± 0.056	10.960 ± 0.066	14.458 ± 0.062	32.424 ± 0.700
45	9.224 ± 0.089	17.572 ± 0.101	19.331 ± 0.086	53.526 ± 1.169
75	16.431 ± 0.152	30.091 ± 0.167	27.525 ± 0.120	103.572 ± 2.146
150	37.105 ± 0.330	63.748 ± 0.333	45.013 ± 0.197	236.810 ± 4.811
230	62.200 ± 0.598	107.936 ± 0.574	60.446 ± 0.277	424.070 ± 8.639
450	134.132 ± 1.183	240.441 ± 1.258	97.679 ± 0.477	913.513 ± 18.345
900	310.763 ± 2.848	547.173 ± 2.810	158.124 ± 0.694	2155.480 ± 41.149
1800	703.895 ± 4.450	1310.990 ± 4.915	263.529 ± 0.953	5438.310 ± 74.901

3d polymers in θ -solvent				
N	$\langle R_g^2(N) \rangle$	$\langle R^2(\langle L \rangle) \rangle$	L_{\max}	$\langle R^2(L_{\max}) \rangle$
3	0.494 ± 0.017	1.000 ± 0.000	2.250 ± 0.043	1.763 ± 0.117
5	0.756 ± 0.021	1.797 ± 0.045	3.540 ± 0.054	2.925 ± 0.185
10	1.502 ± 0.051	3.157 ± 0.081	6.370 ± 0.077	7.118 ± 0.493
20	2.761 ± 0.026	4.721 ± 0.027	10.579 ± 0.042	13.168 ± 0.272
30	3.959 ± 0.038	7.963 ± 0.052	13.981 ± 0.061	19.963 ± 0.437
45	5.591 ± 0.051	11.715 ± 0.081	18.417 ± 0.080	28.601 ± 0.603
75	8.470 ± 0.078	15.817 ± 0.095	25.965 ± 0.118	44.178 ± 0.953
150	15.149 ± 0.143	29.655 ± 0.191	40.689 ± 0.194	80.937 ± 1.787
230	21.303 ± 0.187	42.694 ± 0.273	52.353 ± 0.251	113.870 ± 2.434
450	36.335 ± 0.339	72.887 ± 0.464	78.895 ± 0.363	200.131 ± 4.648
900	63.654 ± 0.561	125.023 ± 0.770	119.231 ± 0.560	344.247 ± 7.636
1800	112.573 ± 0.717	223.027 ± 1.010	181.854 ± 0.602	622.995 ± 9.942

TABLE VII. Conformational statistics of lattice trees II. Values for the critical exponents ν and ν_{path} , obtained from best fits of model functions with ($\Delta > 0$) and without ($\Delta = 0$) correction-to-scaling term (see Sec. III D 2, for details) to the corresponding numerical data reported in Table VI. Final estimates with systematic and statistical errors are highlighted in boldface.

2d ideal polymers			
Observable	$\langle R_g^2(N) \rangle$	$\langle R^2(\langle L \rangle) \rangle$	$\langle R^2(L_{\text{max}}) \rangle$
Δ	0.223 ± 0.187	—	—
DOF	6	—	—
$\tilde{\chi}^2$	0.885	—	—
\mathcal{Q}	0.505	—	—
Exponent	$\nu = 0.209 \pm 0.063$	—	—
Δ	0	0	0
DOF	1	1	1
$\tilde{\chi}^2$	0.364	2.039	0.037
\mathcal{Q}	0.546	0.153	0.848
Exponent	$\nu = 0.262 \pm 0.003$	$\nu_{\text{path}} = 0.491 \pm 0.010$	$\nu_{\text{path}} = 0.530 \pm 0.004$
Average	$\nu = \mathbf{0.236 \pm 0.026 \pm 0.063}$	$\nu_{\text{path}} = \mathbf{0.510 \pm 0.020 \pm 0.010}$	
2d polymers in good solvent			
Observable	$\langle R_g^2(N) \rangle$	$\langle R^2(\langle L \rangle) \rangle$	$\langle R^2(L_{\text{max}}) \rangle$
Δ	1.072 ± 0.542	—	—
DOF	6	—	—
$\tilde{\chi}^2$	2.233	—	—
\mathcal{Q}	0.037	—	—
Exponent	$\nu = 0.628 \pm 0.003$	—	—
Δ	0	0	0
DOF	1	1	1
$\tilde{\chi}^2$	1.297	0.093	0.068
\mathcal{Q}	0.255	0.761	0.795
Exponent	$\nu = 0.620 \pm 0.003$	$\nu_{\text{path}} = 0.845 \pm 0.001$	$\nu_{\text{path}} = 0.827 \pm 0.002$
Average	$\nu = \mathbf{0.624 \pm 0.004 \pm 0.003}$	$\nu_{\text{path}} = \mathbf{0.836 \pm 0.009 \pm 0.002}$	
2d polymers in θ -solvent			
Observable	$\langle R_g^2(N) \rangle$	$\langle R^2(\langle L \rangle) \rangle$	$\langle R^2(L_{\text{max}}) \rangle$
Δ	0.166 ± 0.371	—	0.699 ± 7.988
DOF	6	—	6
$\tilde{\chi}^2$	1.415	—	2.039
\mathcal{Q}	0.204	—	0.057
Exponent	$\nu = 0.630 \pm 0.102$	—	$\nu_{\text{path}} = 0.878 \pm 0.039$
Δ	0	0	0
DOF	1	1	1
$\tilde{\chi}^2$	1.127	0.953	0.105
\mathcal{Q}	0.288	0.329	0.746
Exponent	$\nu = 0.597 \pm 0.003$	$\nu_{\text{path}} = 0.869 \pm 0.002$	$\nu_{\text{path}} = 0.900 \pm 0.003$
Average	$\nu = \mathbf{0.614 \pm 0.016 \pm 0.102}$	$\nu_{\text{path}} = \mathbf{0.882 \pm 0.015 \pm 0.039}$	
3d polymers in θ -solvent			
Observable	$\langle R_g^2(N) \rangle$	$\langle R^2(\langle L \rangle) \rangle$	$\langle R^2(L_{\text{max}}) \rangle$
Δ	1.002 ± 0.268	—	1.054 ± 0.885
DOF	6	—	4
$\tilde{\chi}^2$	0.657	—	0.175
\mathcal{Q}	0.685	—	0.951
Exponent	$\nu = 0.403 \pm 0.002$	—	$\nu_{\text{path}} = 0.691 \pm 0.015$
Δ	0	0	0
DOF	1	1	1
$\tilde{\chi}^2$	0.203	5.140	0.529
\mathcal{Q}	0.652	0.023	0.467
Exponent	$\nu = 0.408 \pm 0.001$	$\nu_{\text{path}} = 0.685 \pm 0.008$	$\nu_{\text{path}} = 0.682 \pm 0.009$
Average	$\nu = \mathbf{0.405 \pm 0.003 \pm 0.002}$	$\nu_{\text{path}} = \mathbf{0.686 \pm 0.004 \pm 0.015}$	

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TABLE VIII. Effective (finite tree weight N) and extrapolated ($N \rightarrow \infty$) exponents (θ_ℓ, t_ℓ). The first were obtained by best fits of the Redner-des Cloizeaux function (Eq. (13)) to the numerical distribution functions, $p_N(\ell)$, of linear paths of length ℓ for tree polymers of total weight N . The second were estimated by best fits of the first to suitably (see Sec. III D 3) chosen functions with three (free Δ) and two ($\Delta \equiv 1$) free parameters. The range where each fit was performed and its statistical significance are given by, respectively, N_{\min} , $\tilde{\chi}^2$ and \mathcal{Q} , while final estimates with corresponding systematic and statistical errors are highlighted in boldface.

2d ideal polymers					2d polymers in good solvent				
N	θ_ℓ	t_ℓ	θ_ℓ	t_ℓ	N_{\min}	$\tilde{\chi}^2$	\mathcal{Q}	Exponent	
20	0.389 ± 0.025	3.462 ± 0.162	0.217 ± 0.024	2.859 ± 0.133	20	0.423	0.864	1.017 ± 0.023	2.062 ± 0.024
30	0.447 ± 0.015	3.188 ± 0.080	0.244 ± 0.014	2.704 ± 0.073	20	1.693	0.118	2.093 ± 0.001	2.468 ± 0.022
45	0.507 ± 0.011	2.977 ± 0.053	0.265 ± 0.008	2.678 ± 0.044	20	6.583	0.201	0.901 ± 0.004	2.154 ± 0.006
75	0.573 ± 0.008	2.720 ± 0.031	0.277 ± 0.006	2.622 ± 0.033	230	0.002	0	0.360 ± 0.041	2.091 ± 0.035
150	0.662 ± 0.007	2.520 ± 0.023	0.291 ± 0.004	2.541 ± 0.021	20	9.168	10.253	0.360 ± 0.041	2.091 ± 0.035
230	0.704 ± 0.006	2.460 ± 0.018	0.292 ± 0.004	2.532 ± 0.019	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
450	0.780 ± 0.006	2.283 ± 0.014	0.287 ± 0.003	2.525 ± 0.016	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
900	0.827 ± 0.004	2.236 ± 0.009	0.295 ± 0.002	2.522 ± 0.011	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
1800	0.870 ± 0.003	2.188 ± 0.006	0.292 ± 0.002	2.482 ± 0.008	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
N_{\min}	20	20	20	20	230	0.002	0	0.360 ± 0.041	2.091 ± 0.035
Δ	0.408 ± 0.018	0.495 ± 0.075	1.476 ± 0.48	0.594 ± 1.552	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
$\tilde{\chi}^2$	0.423	1.693	0.66	1.12	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
\mathcal{Q}	0.864	0.118	0.682	0.347	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
Exponent	1.017 ± 0.023	2.062 ± 0.024	0.293 ± 0.001	2.468 ± 0.022	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
N_{\min}	450	230	20	20	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
Δ	1	1	1	1	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
$\tilde{\chi}^2$	6.583	1.603	0.907	1.393	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
\mathcal{Q}	0.01	0.201	0.5	0.203	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
Exponent	0.901 ± 0.004	2.154 ± 0.006	0.294 ± 0.001	2.492 ± 0.006	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
Average	0.959 ± 0.058 ± 0.023	2.108 ± 0.046 ± 0.024	0.294 ± 0.001 ± 0.001	2.480 ± 0.012 ± 0.022	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035

2d polymers in θ-solvent					3d polymers in θ-solvent				
N	θ_ℓ	t_ℓ	θ_ℓ	t_ℓ	N_{\min}	$\tilde{\chi}^2$	\mathcal{Q}	Exponent	
20	0.370 ± 0.031	3.066 ± 0.165	0.369 ± 0.028	3.254 ± 0.162	230	9.168	10.253	0.360 ± 0.041	2.091 ± 0.035
30	0.415 ± 0.019	2.803 ± 0.087	0.439 ± 0.016	2.940 ± 0.077	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
45	0.439 ± 0.011	2.691 ± 0.046	0.475 ± 0.011	2.790 ± 0.046	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
75	0.450 ± 0.009	2.575 ± 0.038	0.522 ± 0.006	2.606 ± 0.024	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
150	0.449 ± 0.008	2.425 ± 0.028	0.570 ± 0.004	2.431 ± 0.013	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
230	0.460 ± 0.005	2.329 ± 0.019	0.599 ± 0.003	2.415 ± 0.009	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
450	0.442 ± 0.004	2.250 ± 0.013	0.621 ± 0.002	2.389 ± 0.006	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
900	0.405 ± 0.004	2.342 ± 0.017	0.644 ± 0.002	2.334 ± 0.005	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
1800	0.403 ± 0.002	2.173 ± 0.008	0.658 ± 0.001	2.263 ± 0.003	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
N_{\min}	230	20	20	20	230	9.168	10.253	0.360 ± 0.041	2.091 ± 0.035
Δ	0.408 ± 0.147	0.463 ± 0.159	0.562 ± 0.011	0.285 ± 0.102	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
$\tilde{\chi}^2$	9.168	10.253	0.665	12.004	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
\mathcal{Q}	0.002	0	0.678	0	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
Exponent	0.360 ± 0.041	2.091 ± 0.035	0.689 ± 0.005	2.069 ± 0.05	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
N_{\min}	450	20	450	450	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
Δ	1	1	1	1	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
$\tilde{\chi}^2$	4.434	14.918	0.674	24.532	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
\mathcal{Q}	0.035	0	0.412	0	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
Exponent	0.389 ± 0.003	2.206 ± 0.006	0.671 ± 0.002	2.227 ± 0.004	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035
Average	0.375 ± 0.015 ± 0.041	2.149 ± 0.057 ± 0.035	0.680 ± 0.009 ± 0.005	2.148 ± 0.079 ± 0.050	20	0.002	0	0.360 ± 0.041	2.091 ± 0.035

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TABLE IX. Effective exponents ($\theta_{\text{path}}, t_{\text{path}}$) obtained by best fits of the Redner-des Cloizeaux function (Eq. (13)) to the numerical distribution functions, $p_N(\vec{r})$, of end-to-end spatial vectors \vec{r} between nodes of tree polymers of total weight N . Final estimates with corresponding systematic and statistical errors are given in boldface.

		2d ideal polymers		2d polymers in good solvent	
N	ℓ	θ_{path}	t_{path}	θ_{path}	t_{path}
450	16	-0.078 ± 0.008	2.196 ± 0.012	1.653 ± 0.033	7.518 ± 0.179
450	32	0.002 ± 0.003	2.020 ± 0.005	1.802 ± 0.015	7.375 ± 0.073
900	16	-0.015 ± 0.010	2.102 ± 0.013	1.651 ± 0.033	7.594 ± 0.183
900	32	-0.008 ± 0.005	2.057 ± 0.007	1.793 ± 0.012	7.555 ± 0.058
1800	16	0.012 ± 0.014	2.053 ± 0.017	1.641 ± 0.032	7.558 ± 0.177
1800	32	0.004 ± 0.004	2.021 ± 0.007	1.820 ± 0.013	7.512 ± 0.063
1800	64	0.004 ± 0.001	2.022 ± 0.003	1.851 ± 0.006	7.515 ± 0.028
Average		$-0.011 \pm 0.011 \pm 0.014$	$2.067 \pm 0.022 \pm 0.017$	$1.744 \pm 0.032 \pm 0.033$	$7.518 \pm 0.024 \pm 0.183$

		2d polymers in θ -solvent		3d polymers in θ -solvent	
N	ℓ	θ_{path}	t_{path}	θ_{path}	t_{path}
450	16	1.067 ± 0.019	5.402 ± 0.079	0.509 ± 0.009	3.153 ± 0.018
450	32	1.202 ± 0.012	5.929 ± 0.054	0.514 ± 0.006	2.966 ± 0.012
900	16	1.071 ± 0.021	5.482 ± 0.091	0.505 ± 0.015	3.173 ± 0.030
900	32	1.296 ± 0.011	5.775 ± 0.047	0.561 ± 0.003	3.069 ± 0.006
1800	16	1.059 ± 0.021	5.527 ± 0.095	0.500 ± 0.009	3.261 ± 0.019
1800	32	1.263 ± 0.011	5.965 ± 0.050	0.574 ± 0.002	3.096 ± 0.005
1800	64	1.422 ± 0.007	6.198 ± 0.029	0.572 ± 0.002	3.031 ± 0.003
Average		$1.197 \pm 0.049 \pm 0.021$	$5.754 \pm 0.103 \pm 0.095$	$0.533 \pm 0.012 \pm 0.015$	$3.107 \pm 0.034 \pm 0.030$

TABLE X. Effective (finite tree weight N) and extrapolated ($N \rightarrow \infty$) exponents ($\theta_{\text{tree}}, t_{\text{tree}}$). The first were obtained by best fits of the Redner-des Cloizeaux function (Eq. (13)) to the numerical distribution functions, $p_N(\vec{r})$, of end-to-end spatial vectors \vec{r} between nodes of tree polymers of total weight N . As for the second we adopted the same methodology as for exponents (θ_ℓ, t_ℓ), and the notation of the table is the same as in Table VIII. In the case of $2d$ polymers in good solvent, the 3-parameter fit for t_{tree} fails, then errors are calculated based on the ones from ideal trees for which (statistical error for 3-parameter fit) ≈ 2.5 (statistical error of 2-parameter fit) ≈ 4.8 (systematic error).

$2d$ ideal polymers			$2d$ polymers in good solvent	
N	θ_{tree}	t_{tree}	θ_{tree}	t_{tree}
20	0.368 ± 0.124	1.128 ± 0.057	-0.218 ± 0.078	1.512 ± 0.073
30	0.544 ± 0.128	1.048 ± 0.045	-0.364 ± 0.034	1.683 ± 0.047
45	0.272 ± 0.078	1.173 ± 0.034	-0.428 ± 0.015	1.870 ± 0.030
75	0.089 ± 0.037	1.245 ± 0.020	-0.453 ± 0.006	1.923 ± 0.015
150	-0.006 ± 0.016	1.334 ± 0.011	-0.457 ± 0.002	1.991 ± 0.007
230	-0.065 ± 0.008	1.398 ± 0.007	-0.455 ± 0.001	2.030 ± 0.004
450	-0.006 ± 0.006	1.353 ± 0.005	-0.454 ± 0.001	1.997 ± 0.003
900	-0.001 ± 0.004	1.361 ± 0.004	-0.455 ± 0.001	2.085 ± 0.002
1800	0.013 ± 0.002	1.343 ± 0.002	-0.457 ± 0.000	2.113 ± 0.001
N_{min}	150	230	20	—
Δ	0.278 ± 3.133	1.175 ± 0.106	1.626 ± 0.259	—
$\tilde{\chi}^2$	11.770	13.183	4.167	—
Q	$7.7 \cdot 10^{-6}$	$2.8 \cdot 10^{-4}$	$3.4 \cdot 10^{-4}$	—
Exponent	0.079 ± 0.103	1.339 ± 0.006	-0.4560 ± 0.0003	—
N_{min}	230	230	450	450
Δ	1	1	1	1
$\tilde{\chi}^2$	2.709	6.628	2.321	15.579
Q	0.067	0.001	0.128	$7.9 \cdot 10^{-5}$
Exponent	0.024 ± 0.003	1.337 ± 0.002	-0.4574 ± 0.0005	2.154 ± 0.002
Average	$0.051 \pm 0.028 \pm 0.103$	$1.338 \pm 0.001 \pm 0.006$	$-0.4567 \pm 0.0007 \pm 0.0005$	$2.154 \pm 0.010 \pm 0.005$
$2d$ polymers in θ -solvent			$3d$ polymers in θ -solvent	
N	θ_{tree}	t_{tree}	θ_{tree}	t_{tree}
20	-0.181 ± 0.081	1.528 ± 0.063	0.156 ± 0.340	0.977 ± 0.084
30	-0.309 ± 0.054	1.695 ± 0.061	-0.766 ± 0.112	1.462 ± 0.069
45	-0.350 ± 0.027	1.786 ± 0.039	-0.904 ± 0.068	1.663 ± 0.063
75	-0.390 ± 0.010	1.936 ± 0.021	-0.924 ± 0.032	1.861 ± 0.043
150	-0.379 ± 0.004	1.934 ± 0.009	-0.833 ± 0.015	1.848 ± 0.023
230	-0.360 ± 0.002	1.868 ± 0.005	-0.798 ± 0.009	1.878 ± 0.016
450	-0.371 ± 0.001	1.960 ± 0.004	-0.732 ± 0.005	1.848 ± 0.009
900	-0.372 ± 0.001	1.934 ± 0.003	-0.686 ± 0.003	1.814 ± 0.006
1800	-0.376 ± 0.001	1.889 ± 0.002	-0.657 ± 0.002	1.775 ± 0.003
N_{min}	230	230	150	30
Δ	0.895 ± 0.344	7.175 ± 2.844	0.505 ± 0.041	1.531 ± 0.036
$\tilde{\chi}^2$	6.412	368.506	0.473	20.447
Q	0.011	0.000	0.623	$1.8 \cdot 10^{-20}$
Exponent	-0.379 ± 0.003	1.905 ± 0.002	-0.582 ± 0.025	1.788 ± 0.003
N_{min}	230	450	450	450
Δ	1	1	1	1
$\tilde{\chi}^2$	3.228	30.778	1.583	5.065
Q	0.040	$2.9 \cdot 10^{-8}$	0.208	0.024
Exponent	-0.378 ± 0.001	1.867 ± 0.003	-0.631 ± 0.003	1.750 ± 0.005
Average	$-0.3787 \pm 0.0003 \pm 0.0029$	$1.886 \pm 0.019 \pm 0.003$	$-0.607 \pm 0.024 \pm 0.025$	$1.769 \pm 0.019 \pm 0.005$

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